

# Oxide interfaces

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5<sup>th</sup> International School of Oxide Electronics (ISOE2019)

Cargèse, July 2019

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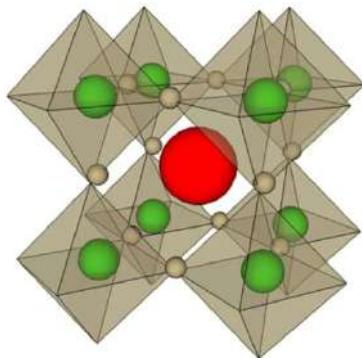
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A. Santander

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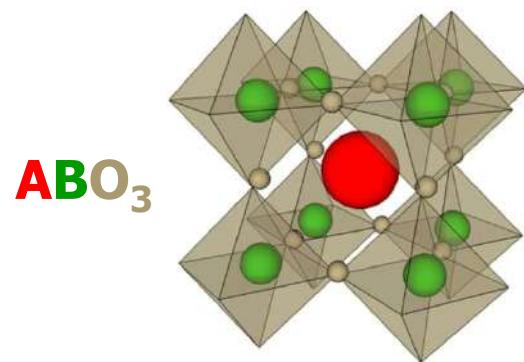
J. Santamaria, M. Varela

## Transition metal perovskite oxides



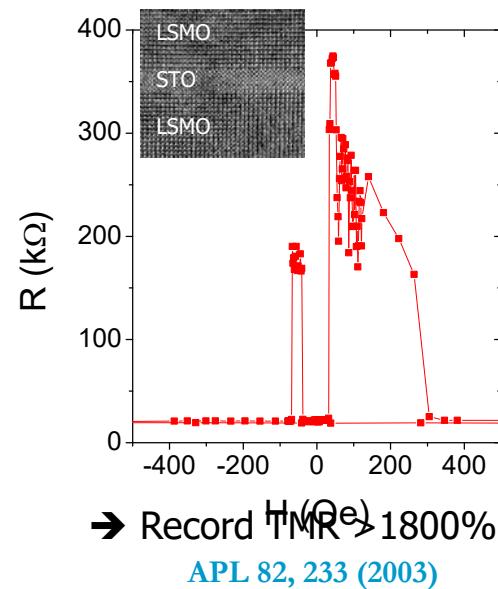
- Very **flexible** structure  
doping, tuning of bond lengths and angles
- Broad range of **electronic states**  
superconductivity, ferroelectricity, magnetic order, orbital order
- Competition : **giant/coupled responses**  
colossal magnetoresistance, magnetoelectric coupling
- Multifunctional heteroepitaxial architectures

### Transition metal perovskite oxides

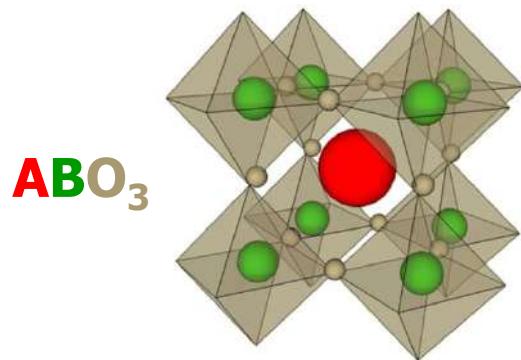


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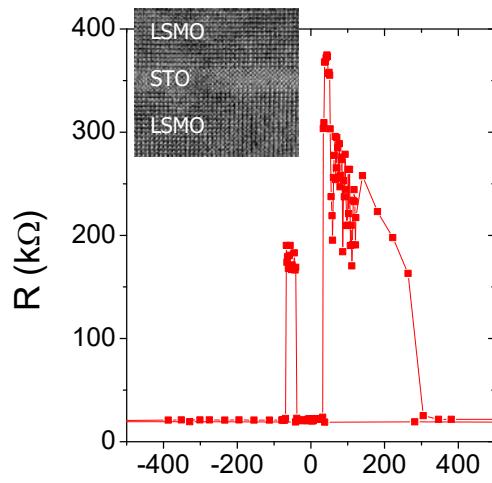
#### MTJs based on $\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3$



### Transition metal perovskite oxides



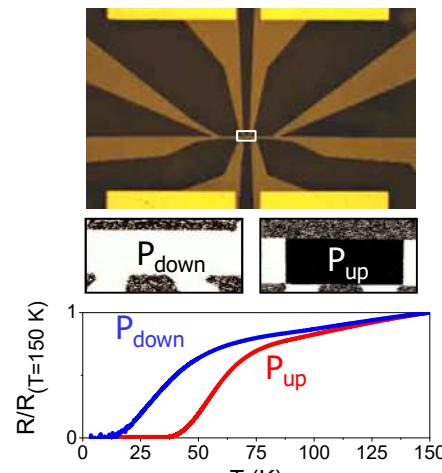
MTJs based on La<sub>2/3</sub>Sr<sub>1/3</sub>MnO<sub>3</sub>



→ Record TMR > 1800%  
APL 82, 233 (2003)

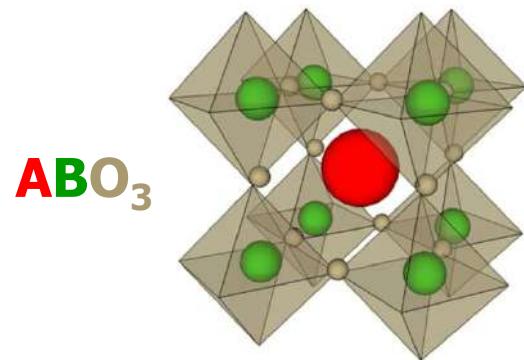
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Ferroelectric FETs based on  
in YBa<sub>2</sub>CuO<sub>7</sub>/BiFeO<sub>3</sub>

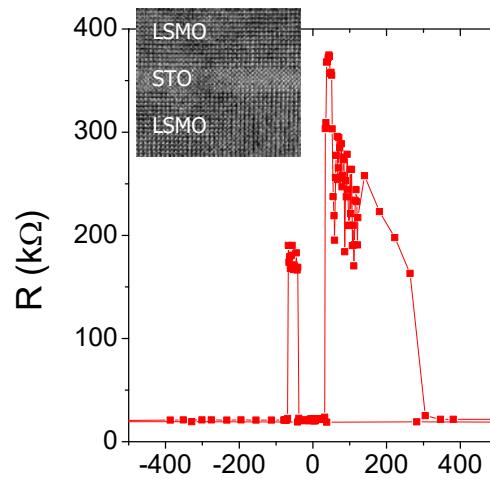


→ Non-volatile control of SC  
PRL 107, 247002 (2011)

## Transition metal perovskite oxides



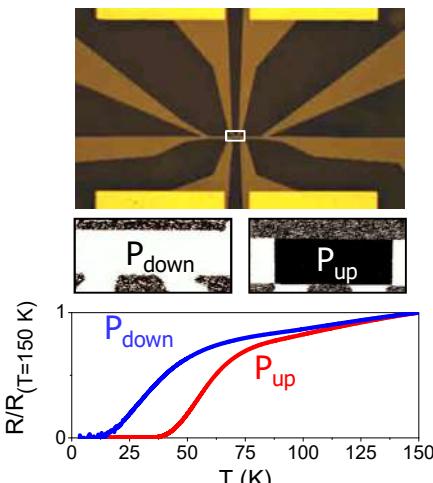
**MTJs based on La<sub>2/3</sub>Sr<sub>1/3</sub>MnO<sub>3</sub>**



→ Record H<sub>MR</sub> 1800%  
*APL 82, 233 (2003)*

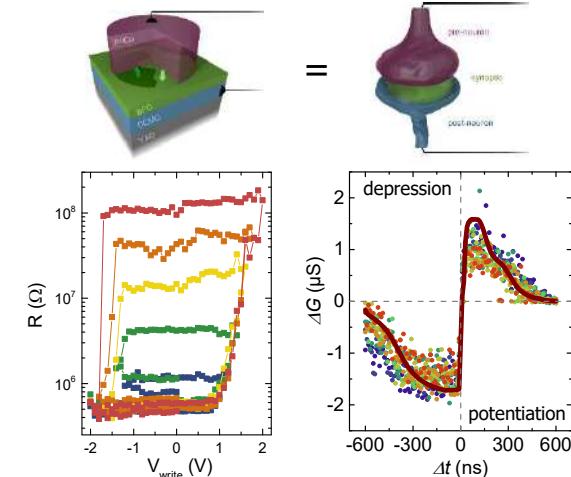
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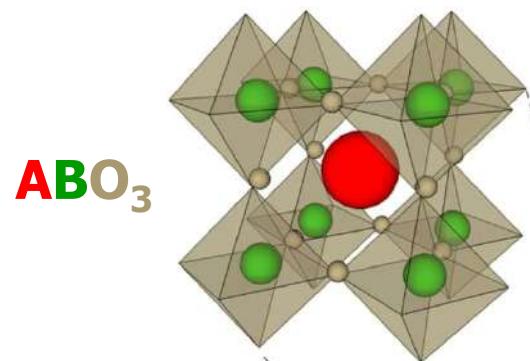
→ Non-volatile control of SC  
*PRL 107, 247002 (2011)*

**Electronic synapses based on  
(Ca,Ce)MnO<sub>3</sub>/BiFeO<sub>3</sub> FTJs**

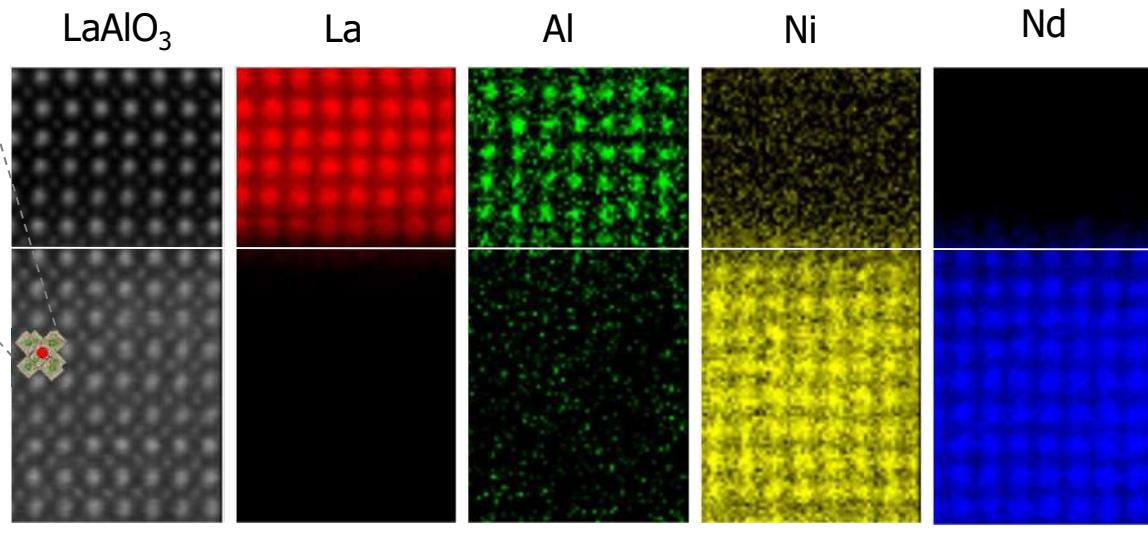


→ Memristive response + learning  
*Nature 460, 81 (2009) ; Nature Mater. 11, 860 (2012) ; Nature Comm. 8, 14736 (2017)*

## Transition metal perovskite oxides

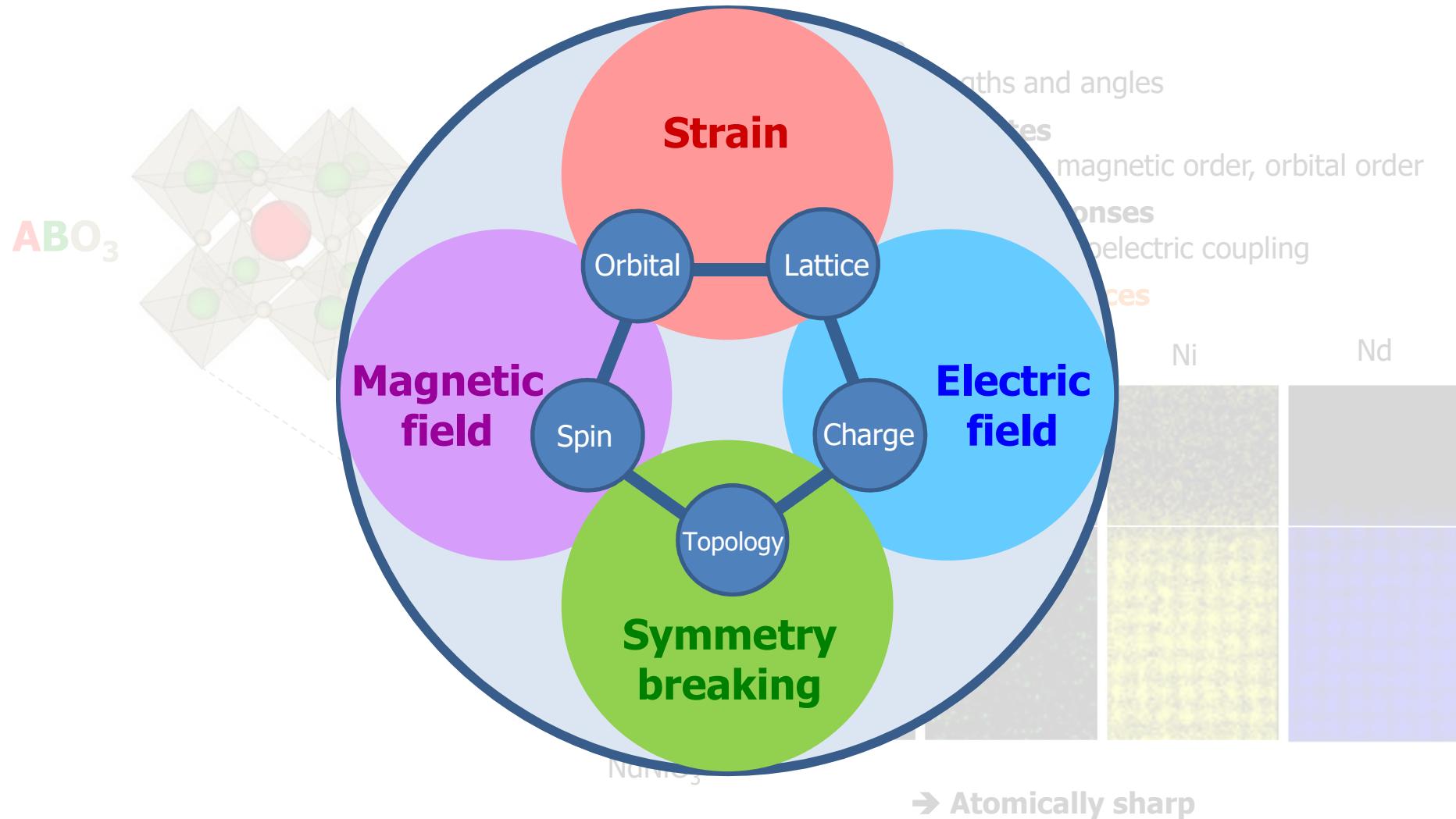


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- Competition : **giant/coupled responses**  
colossal magnetoresistance, magnetoelectric coupling
- Multifunctional heteroepitaxial architectures
- **New interface properties**



→ Atomically sharp

# Oxide interfaces : new playground for physicists



STEM by X. Yang & A. Gloter (LPS Orsay)

### **1. $d_0$ -oxide-based interfaces**

#### **1.1 Physics of bulk SrTiO<sub>3</sub>**

#### **1.2 LaAlO<sub>3</sub>/SrTiO<sub>3</sub> 2DEGs**

#### **1.3 Other SrTiO<sub>3</sub> 2DEGs**

#### **1.4 2DEGs not based on SrTiO<sub>3</sub>**

### **2. Interfaces between oxides with partially filled d shells**

#### **2.1 « Correlated » oxide perovskites**

#### **2.2 Nickelate/Titanate interfaces**

# 1. $d_0$ -oxide-based interfaces

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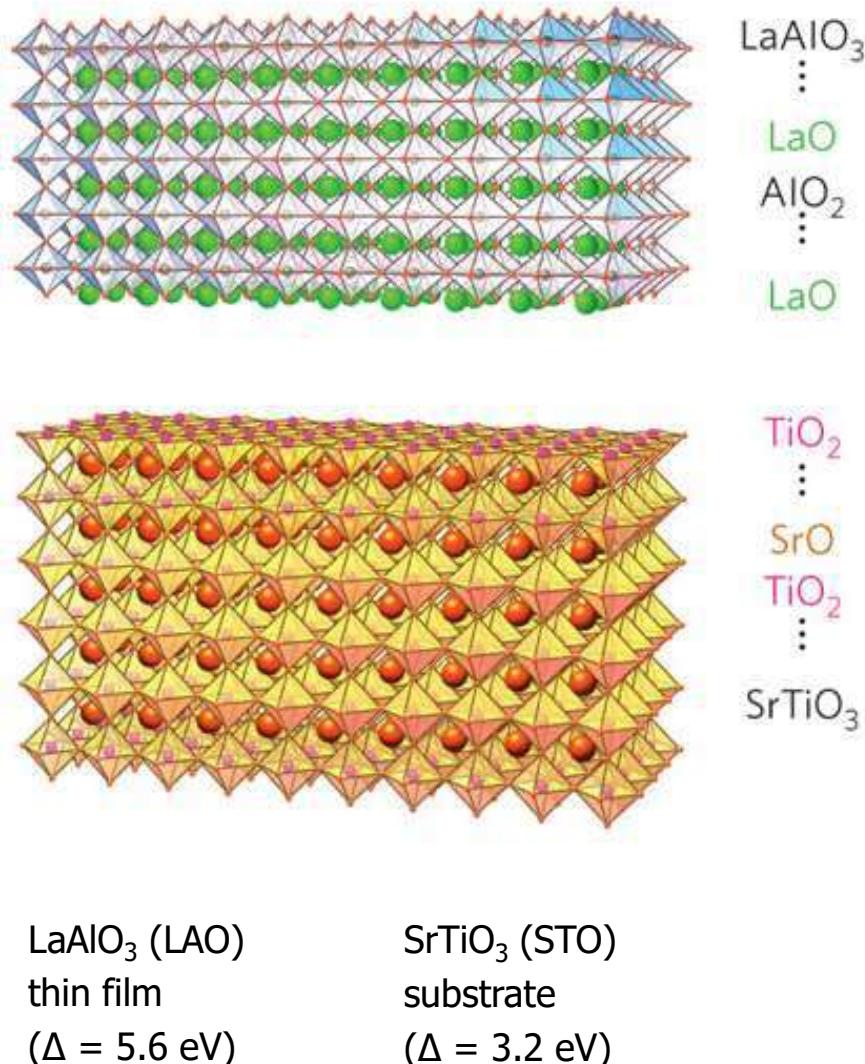
## 1.4 2DEGs not based on $\text{SrTiO}_3$

# 2. Interfaces between oxides with partially filled d shells

## 2.1 « Correlated » oxide perovskites

## 2.2 Nickelate/Titanate interfaces

## An unexpected discovery



## A high-mobility electron gas at the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> heterointerface

A. Ohtomo<sup>1,2,3</sup> & H. Y. Hwang<sup>1,3,4</sup>

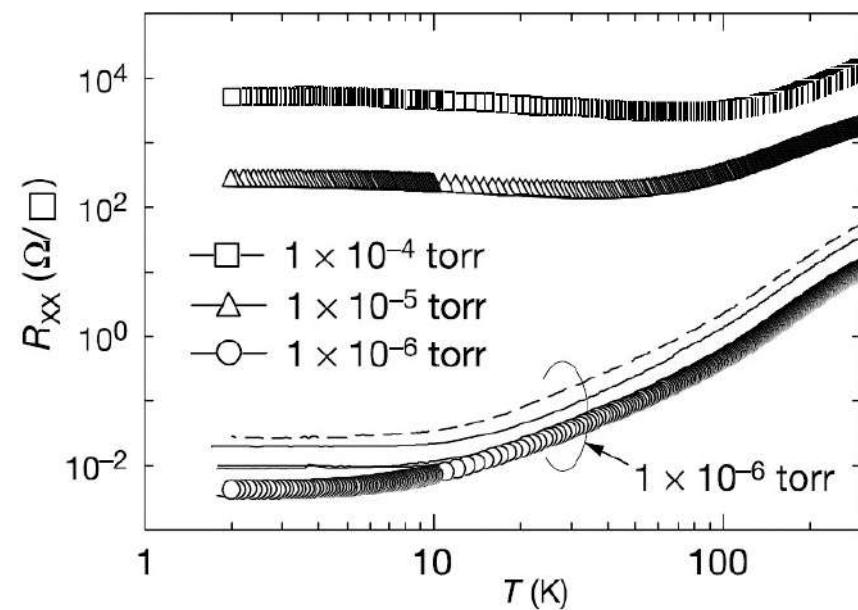
<sup>1</sup>Bell Laboratories, Lucent Technologies, Murray Hill, New Jersey 07974, USA

<sup>2</sup>Institute for Materials Research, Tohoku University, Sendai, 980-8577, Japan

<sup>3</sup>Japan Science and Technology Agency, Kawaguchi, 332-0012, Japan

<sup>4</sup>Department of Advanced Materials Science, University of Tokyo, Kashiwa, Chiba, 277-8651, Japan

LaAlO<sub>3</sub> (60 Å) /SrTiO<sub>3</sub>(001)



Ohtomo & Hwang, Nature 427, 423 (2004)

# 1. $d_0$ -oxide-based interfaces

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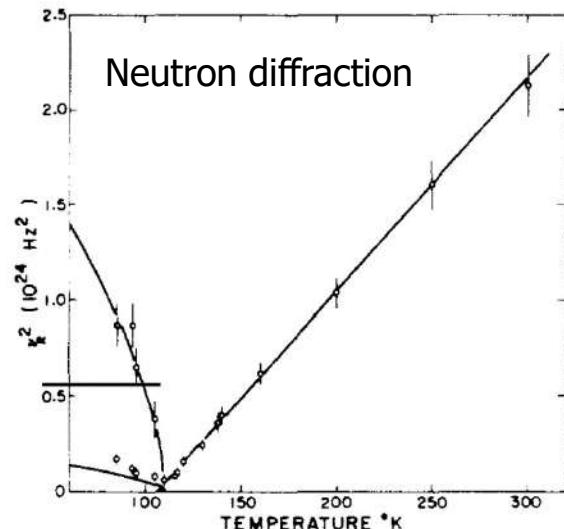
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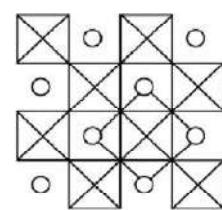
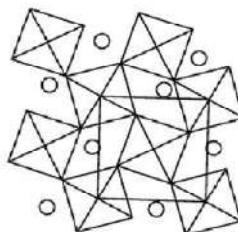
## 2.1 « Correlated » oxide perovskites

## 2.2 Nickelate/Titanate interfaces

## Structural properties

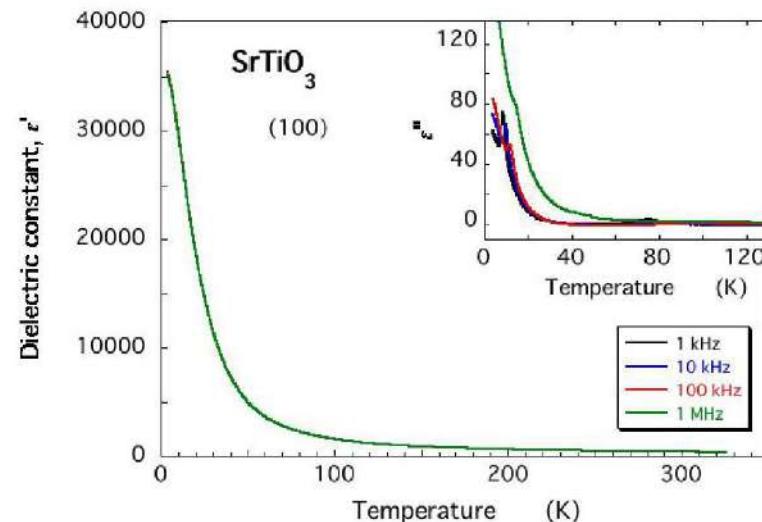


Cowley et al, Solid. State. Commun. 7, 181 (1967)



- SrTiO<sub>3</sub> is cubic at room temperature and above 105 K
- Below 105 K, it is tetragonal, with oxygen octahedra tilt pattern  $a^0a^0c$

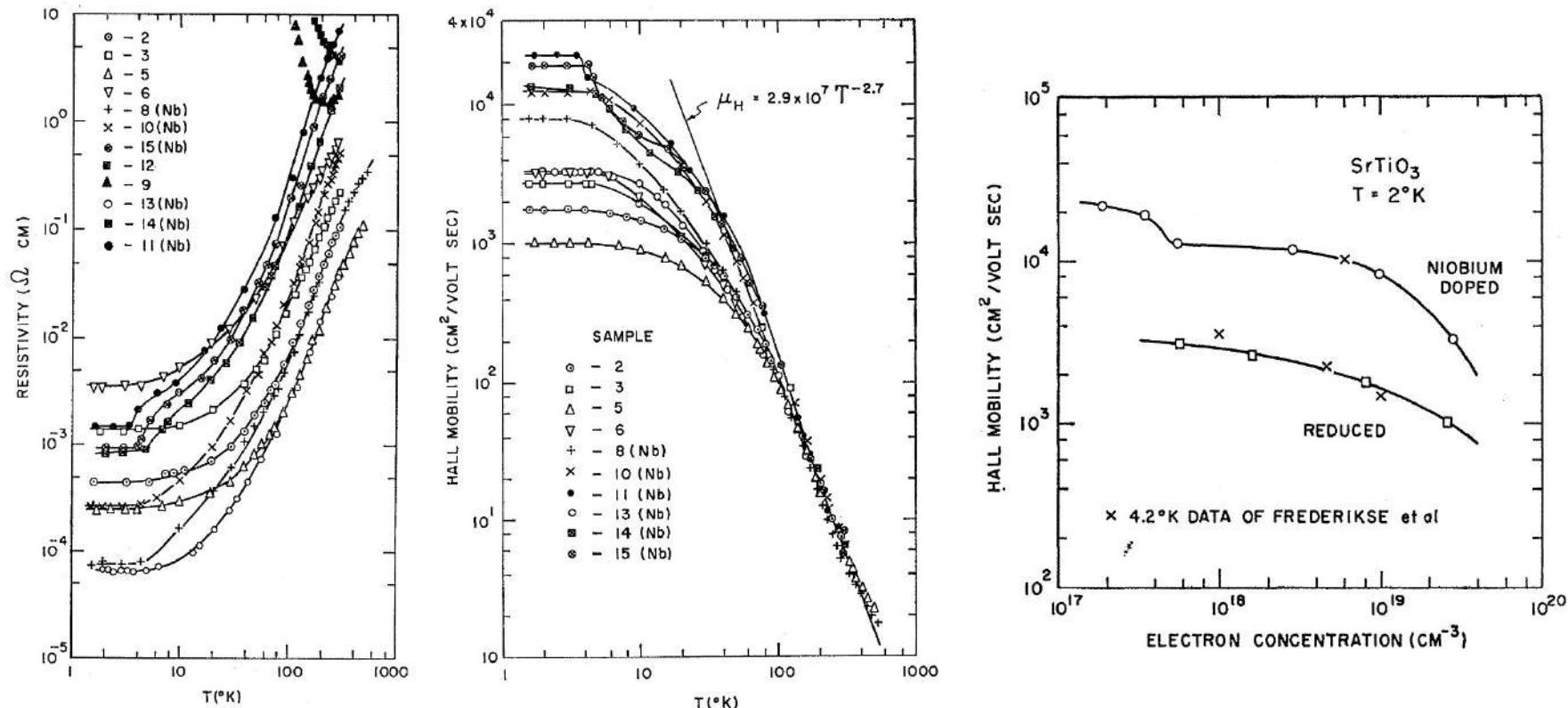
## Dielectric properties



Hideshi et al, JPSJ 85, 045703 (2016)

- SrTiO<sub>3</sub> has a large dielectric constant that diverges at low temperature
- « Quantum paraelectric » : ferroelectric instability suppressed by quantum fluctuations

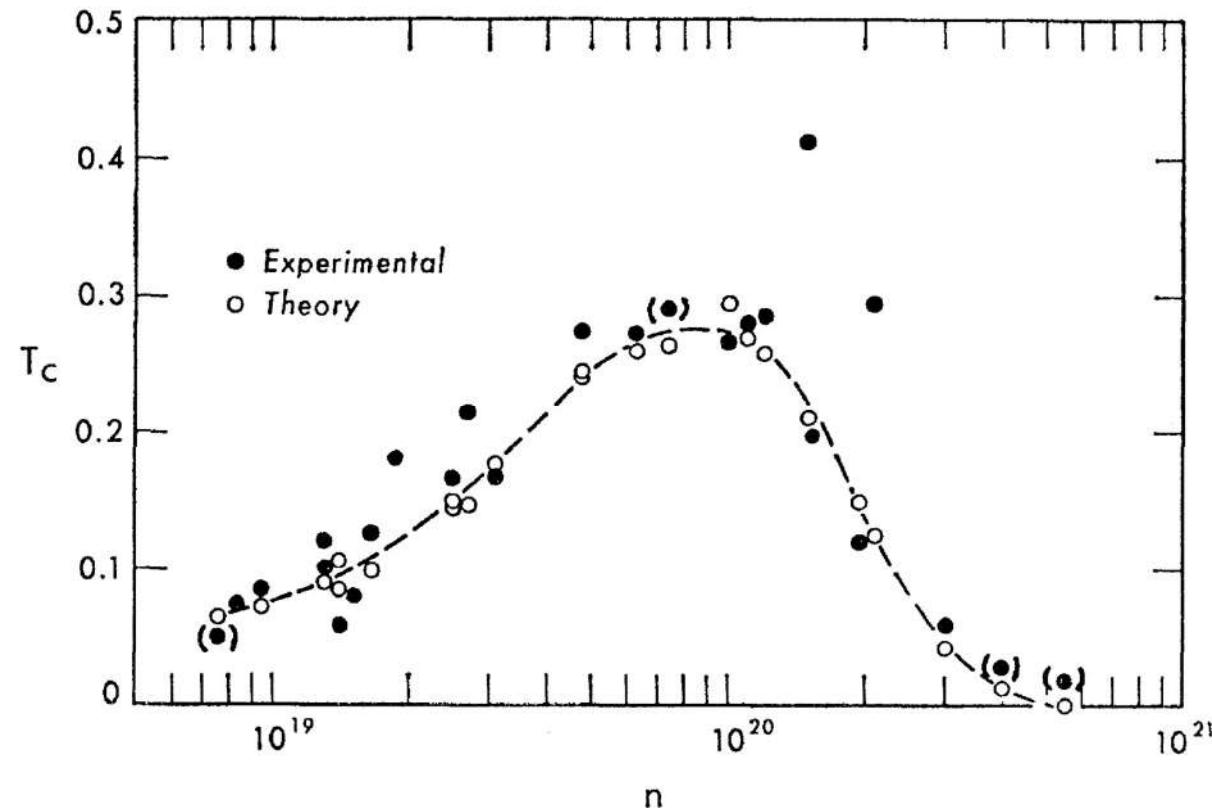
## Transport properties of electron doped SrTiO<sub>3</sub>



- SrTiO<sub>3</sub> can be (n-type) doped into a metal by La substitution at the Sr site, Nb substitution at the Ti site, or by the creation of oxygen vacancies
- Minute doping amounts (e.g. 10 ppm) are enough to induce metallicity
- Electron mobility is very high (>10000 cm<sup>2</sup>/Vs) at low T and decreases with doping

Tufte and Chapman PR 155, 796 (1967)

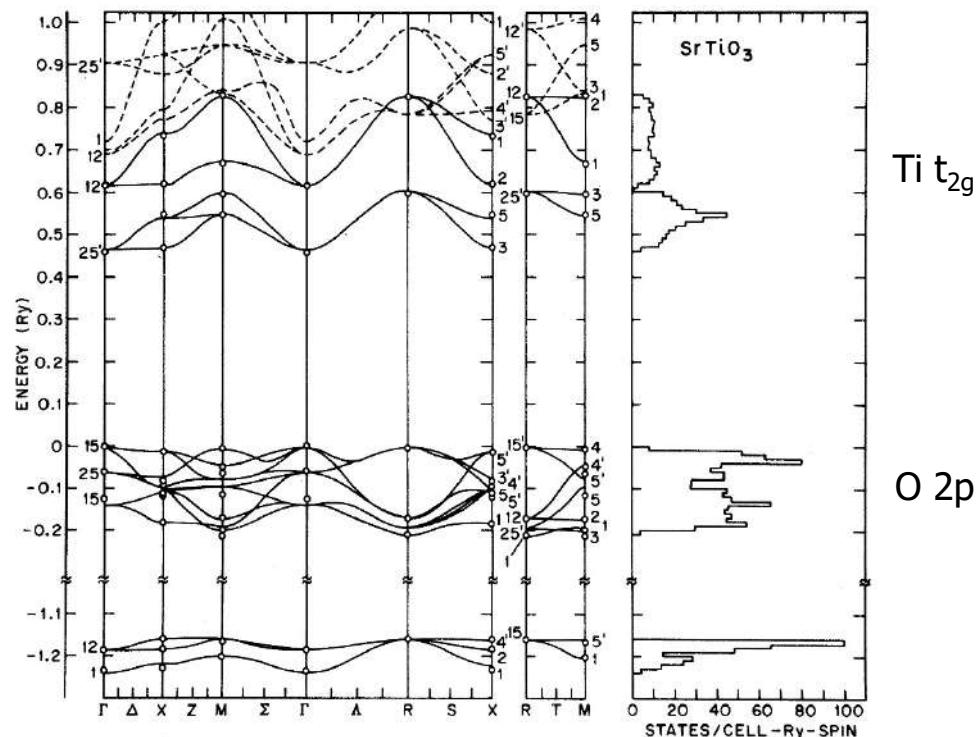
## Superconductivity in electron doped SrTiO<sub>3</sub>



- SrTiO<sub>3</sub> becomes superconducting below about 300 mK for doping levels  $>$  a few  $10^{18} \text{ cm}^{-3}$
- Dome-like phase diagram as in high  $T_c$  superconductors

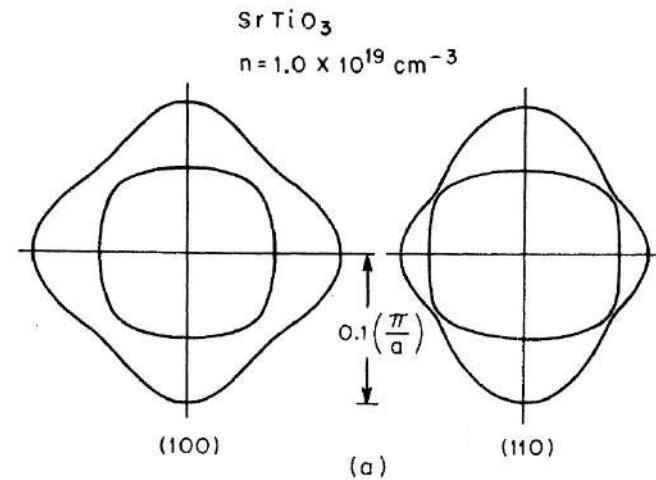
Koonce PR, 163, 780 (1967)

# Electronic structure of SrTiO<sub>3</sub>



Mattheiss PRB 6, 4718 (1972)

Ti  $t_{2g}$   
O  $2p$



- SrTiO<sub>3</sub> is a band insulator, with the valence band made of O 2p states and the conduction band of Ti 3d  $t_{2g}$  states
- The gap is at the Gamma point with two degenerate bands with small and large effective masses
- The light electron band has a « circular » Fermi surface around Gamma while the heavy electron band consists of a double ellipse.

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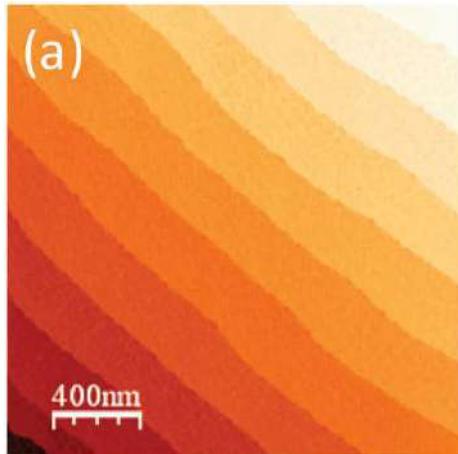
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# How to grow LAO/STO 2DEGs ?

## TiO<sub>2</sub>-terminated substrate



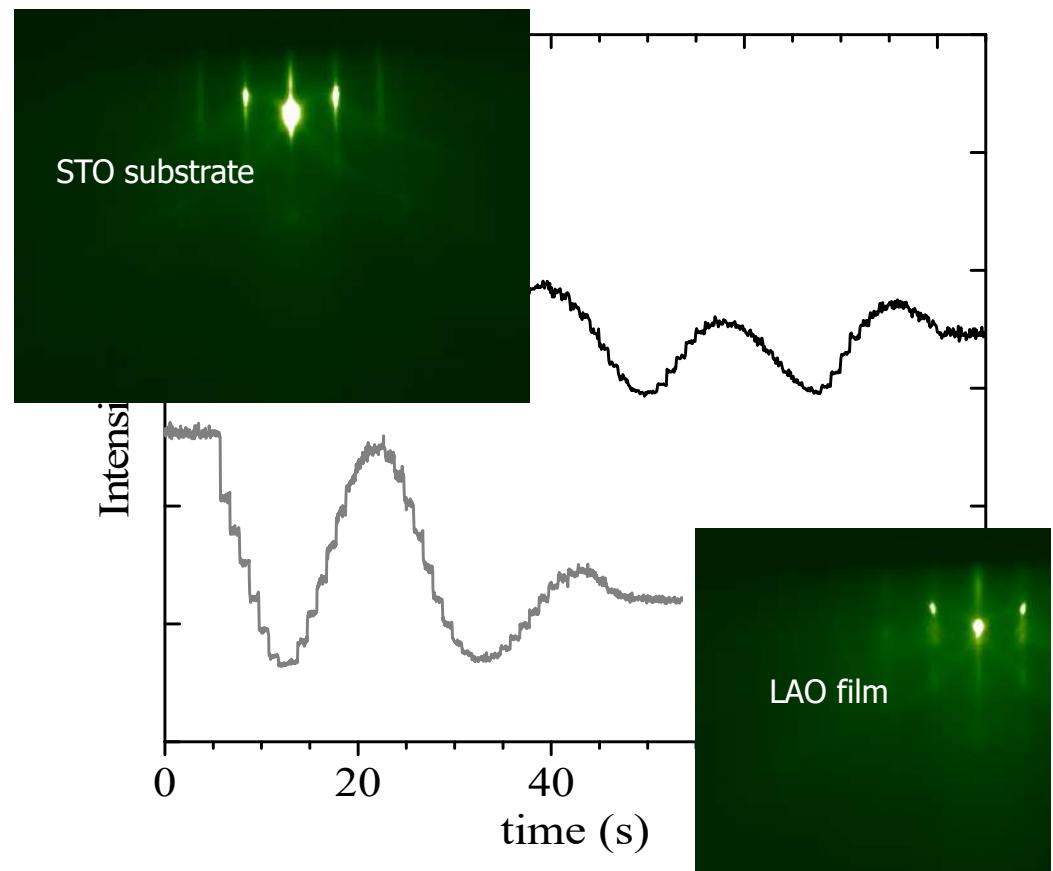
Essential steps:

- Use a TiO<sub>2</sub> terminated STO single crystal
- Grow an integer number of LAO unit cells  $\geq 4$
- Post-anneal in O<sub>2</sub>

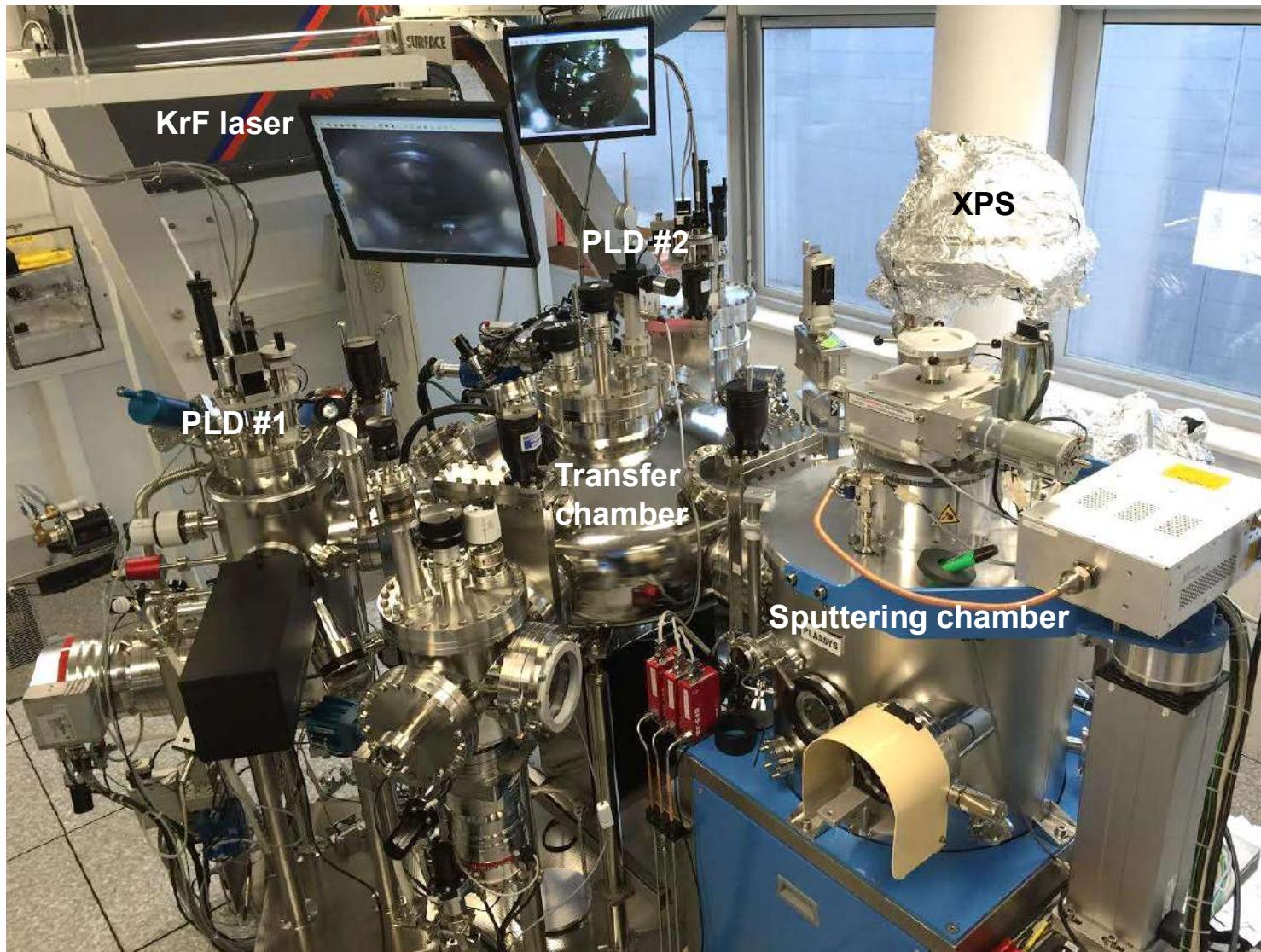
Lesne, MB et al, Nature Comm. 5, 4291 (2014)  
Vaz, MB et al, Adv. Mater. 29, 1700486 (2017)

## PLD growth conditions for LaAlO<sub>3</sub>

- 700-800°C
- $2 \times 10^{-4}$  mbar of O<sub>2</sub>
- KrF excimer (248nm) – 0.6-1.2 J/cm<sup>2</sup> at 1 Hz
- in-situ annealing in high O<sub>2</sub> pressure (0.2-1 bar) at T  $\geq$  500°C for 30-60'

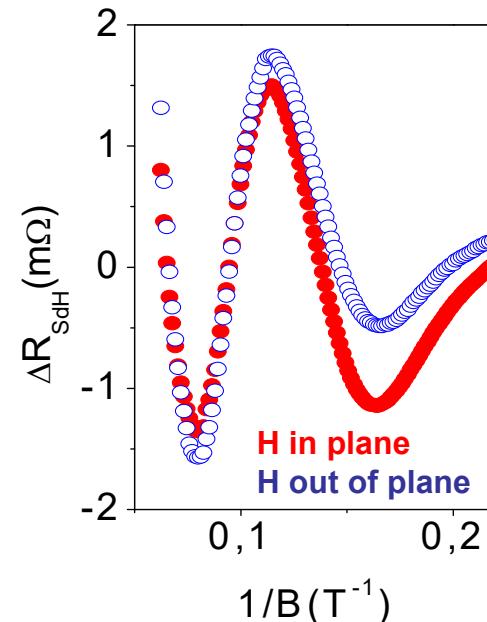
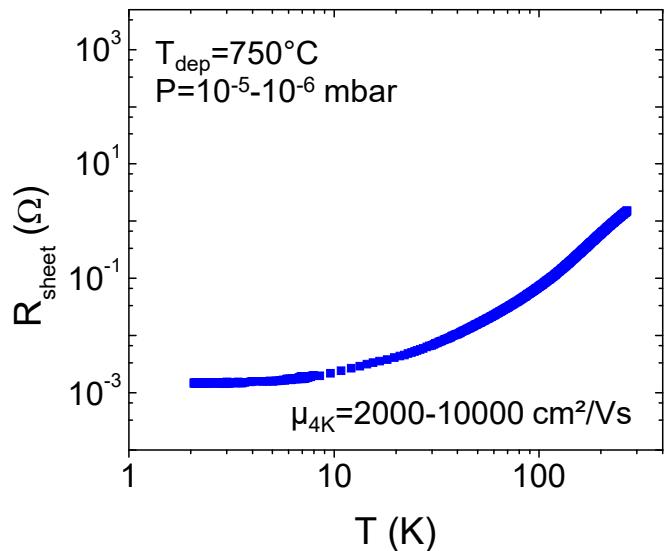


# Coupled PLD+sputtering deposition system



# Do we really have a 2DEG ? How to measure its thickness ?

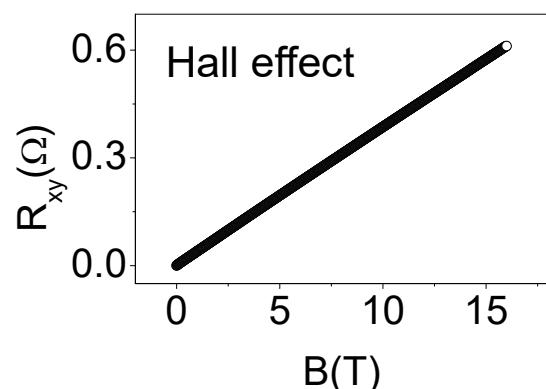
## Magnetotransport measurements



**High-field magnetoresistance:**  
Observation of Shubnikov-de Haas oscillations

$$F_{\text{SdH}} = \frac{\hbar k_F^2}{2e} \quad n = \frac{k_F^3}{3\pi^2}$$

SdH oscillations provide the **carrier density** ( $\text{cm}^{-3}$ )



Hall measurement provide the **sheet carrier density** ( $\text{cm}^{-2}$ )

$$\text{SdH} \rightarrow n \times t_{\text{gas}} = (1/e) B / R_{xy} \leftarrow \text{Hall}$$

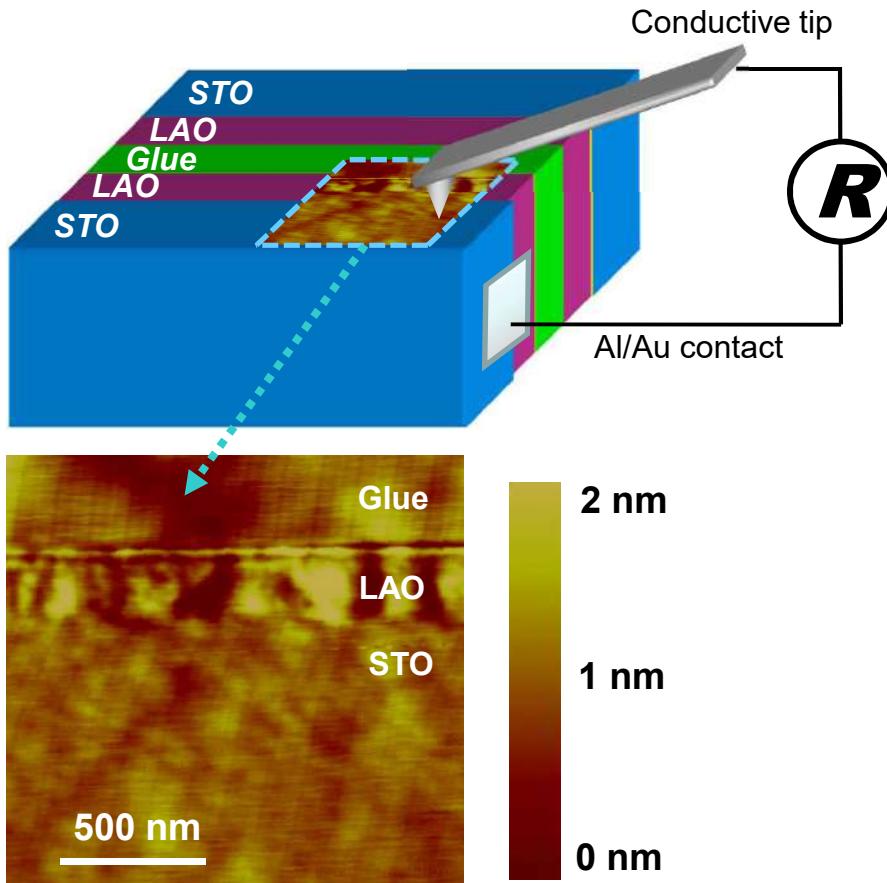
Determination of the electron gas thickness : **~500 μm**

Herranz, MB et al, PRL 98, 216803 (2007)

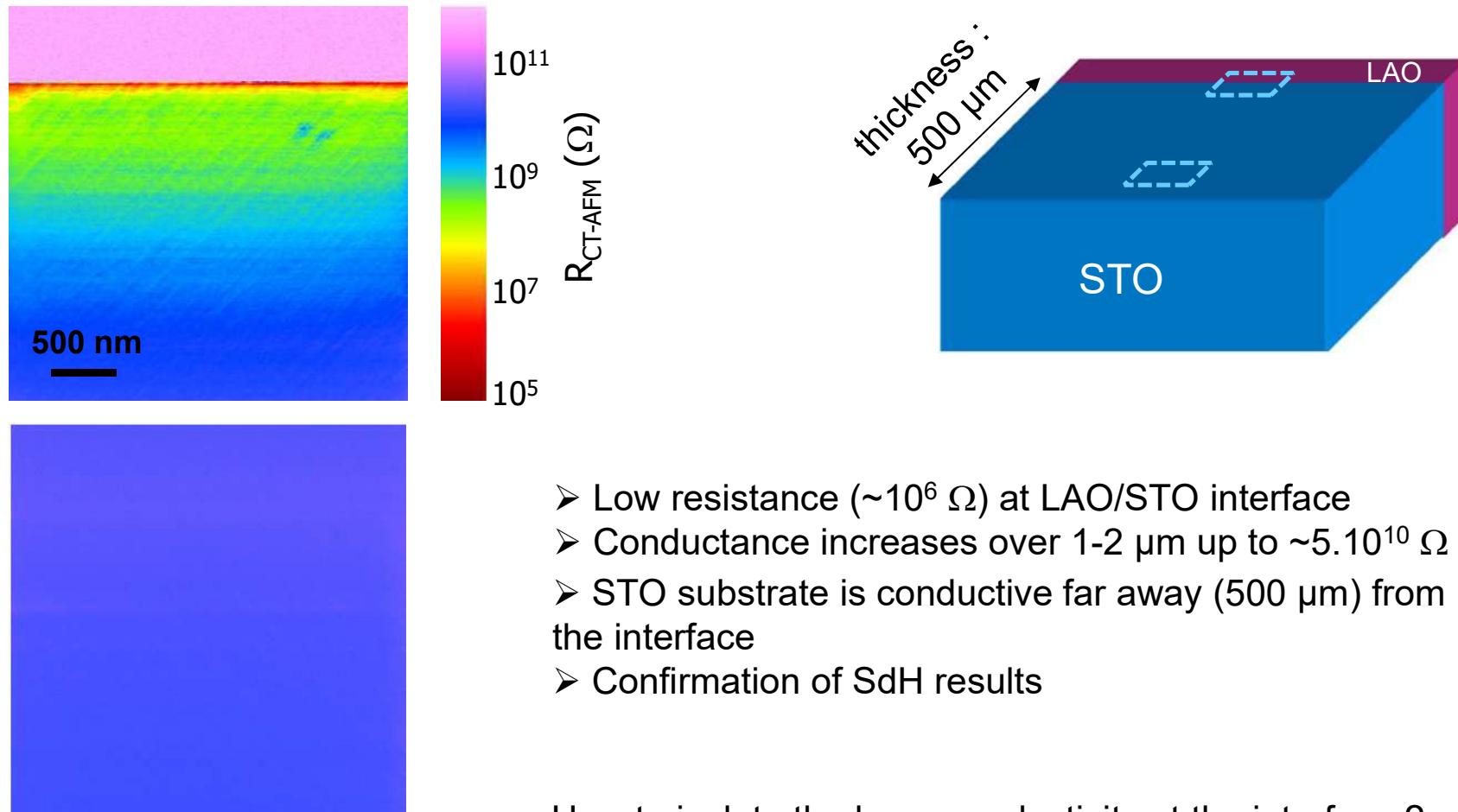
- Metallic behaviour, high mobility, quantum oscillations : nice sample !
- But : electron gas thickness is **500 μm** ! Hint : this sample was **not post-annealed in oxygen**

# Conductive-tip AFM in cross-section samples

## Local measurement of transport properties



## Resistance mapping of non-annealed samples

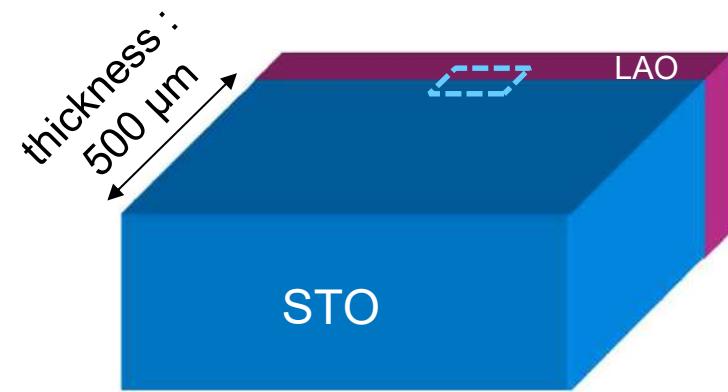
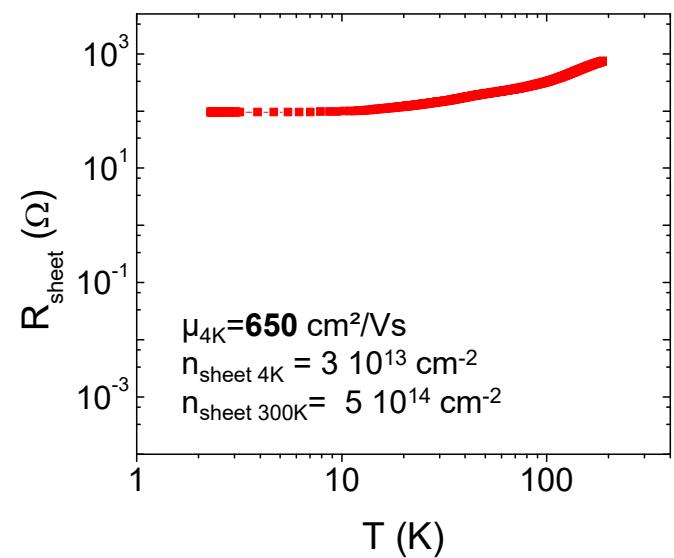


How to isolate the large conductivity at the interface ?  
Get rid of oxygen vacancies : in-situ annealing

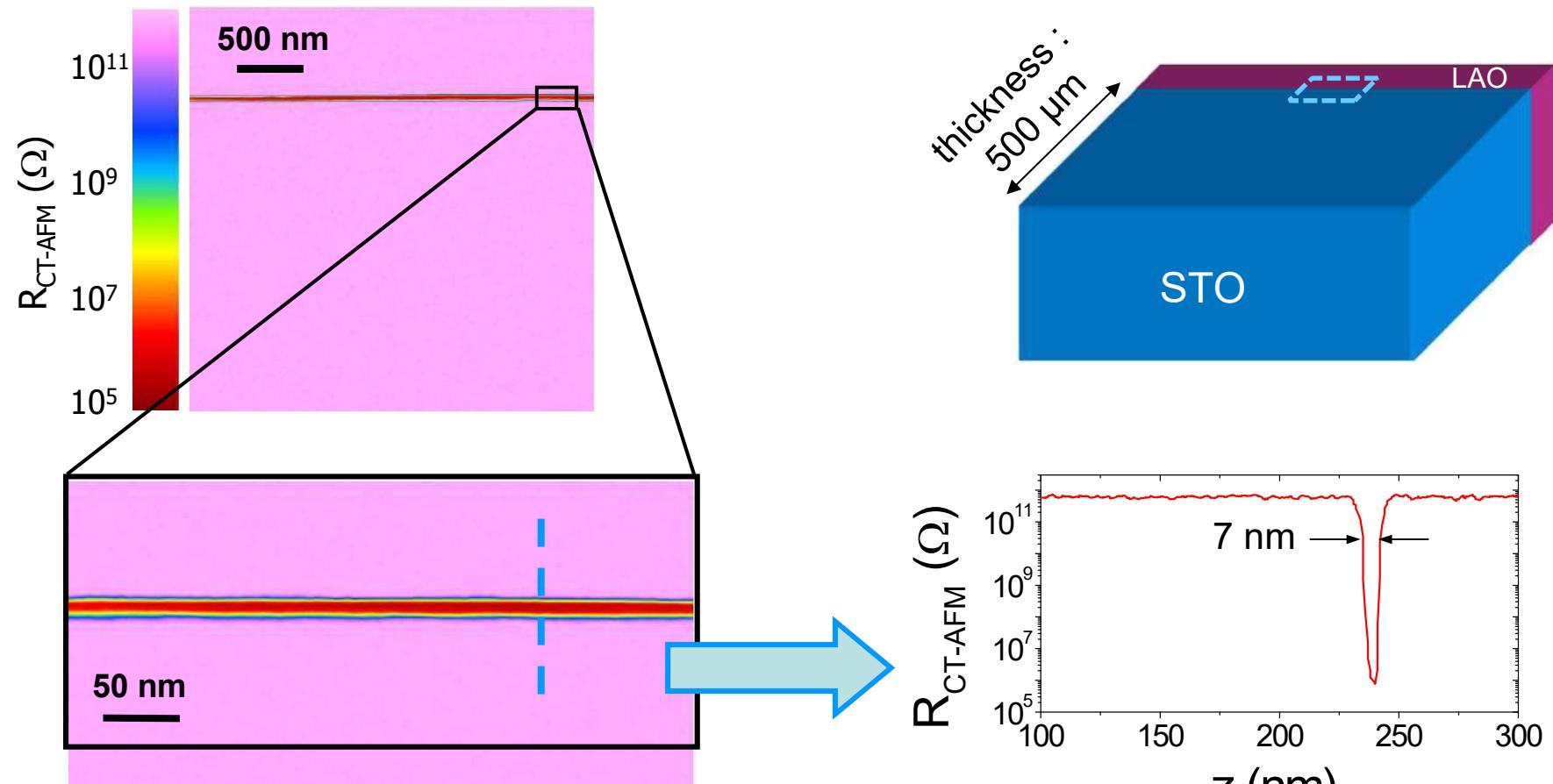
(as in Thiel et al, Science 313, 1942 (2006); Reyren et al, Science 317, 1196 (2007))

Basletic, MB, et al Nature Mater. 7, 621 (2008)

# Resistance mapping of in-situ annealed sample



## Resistance mapping of in-situ annealed sample

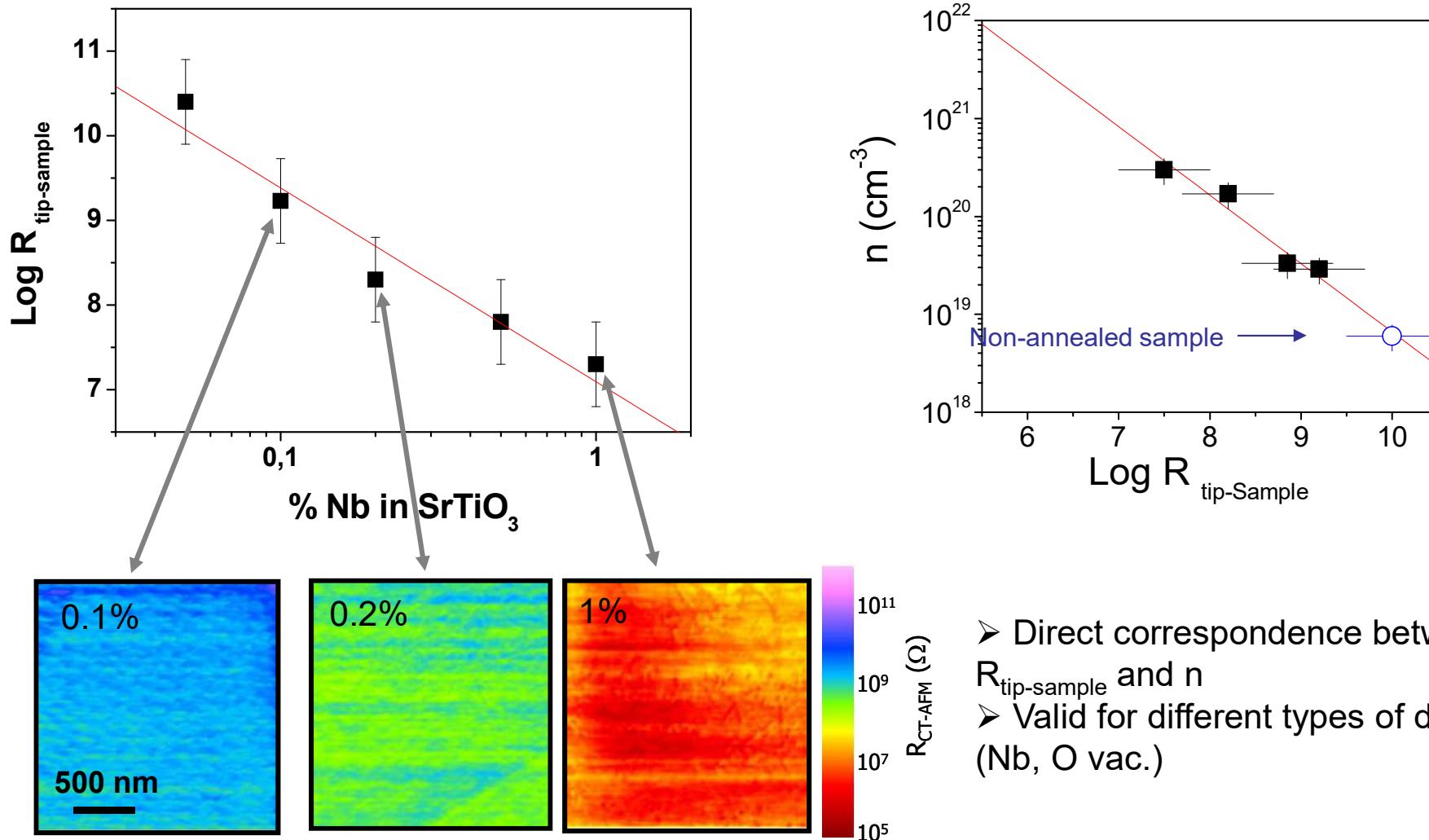


- Low resistance ( $\sim 10^5 \Omega$ ) at LAO/STO interface
- STO substrate is highly insulating far away ( $500 \mu\text{m}$ ) from the interface
- Conduction is **confined** at the interface
- Thickness of the metallic gas : **7 nm** (upper estimate)

Basletic, MB, et al Nature Mater. 7, 621 (2008)

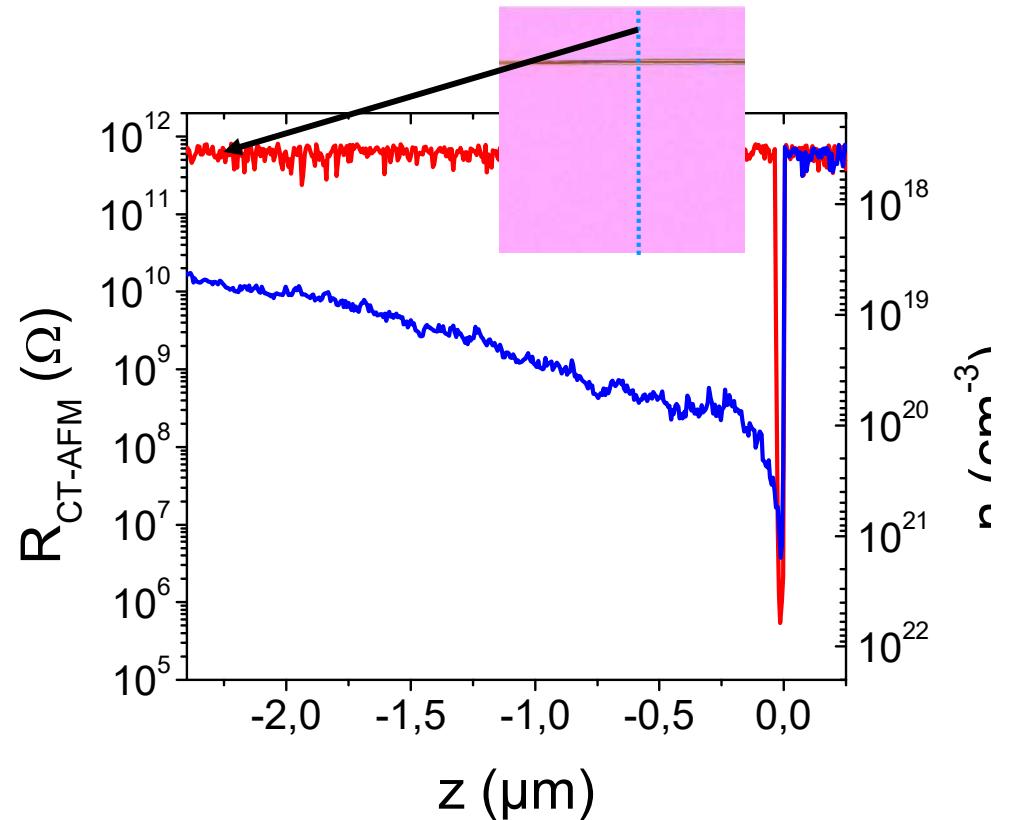
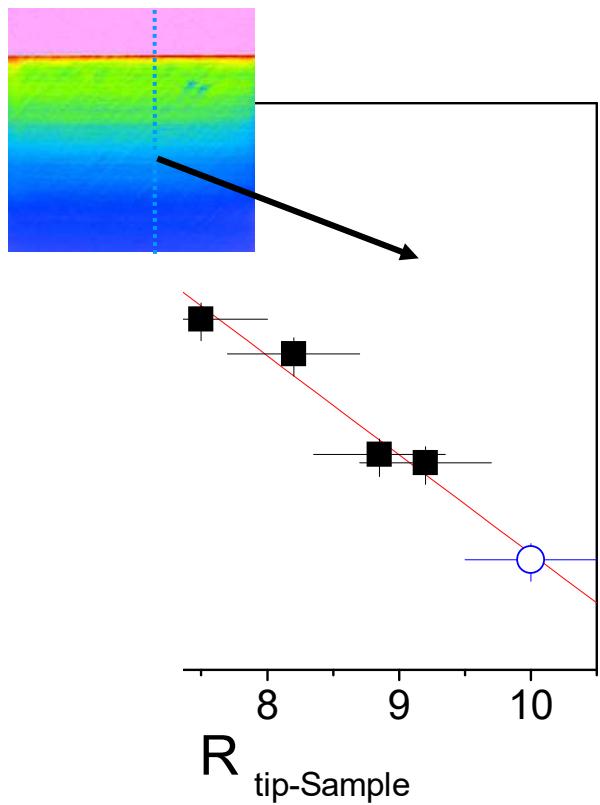
# From resistance to carrier density

## Calibration of $R_{\text{tip-sample}}$ vs $n$ using Nb-doped SrTiO<sub>3</sub> crystals



- Direct correspondence between  $R_{\text{tip-sample}}$  and  $n$
- Valid for different types of dopant (Nb, O vac.)

# Local mapping of the charge carrier distribution



## Non annealed sample

Carrier density away from interface :  $5 \cdot 10^{18} \text{ cm}^{-3}$  :  $t_{\text{gas}} \approx 600 \mu\text{m}$

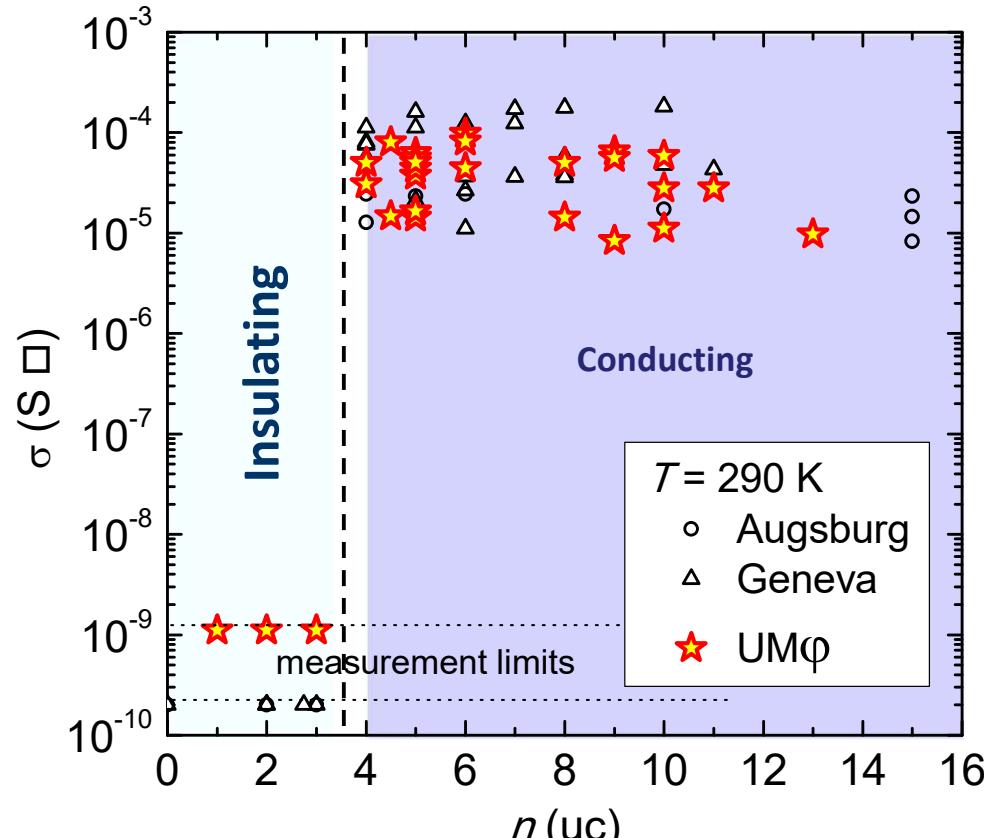
## In-situ annealed sample

Carrier density at the interface :  $7 \cdot 10^{21} \text{ cm}^{-3}$  :  $t_{\text{gas}} \approx 1 \text{ nm}$

Basletic, MB, et al Nature Mater. 7, 621 (2008)

# Critical thickness for conductivity

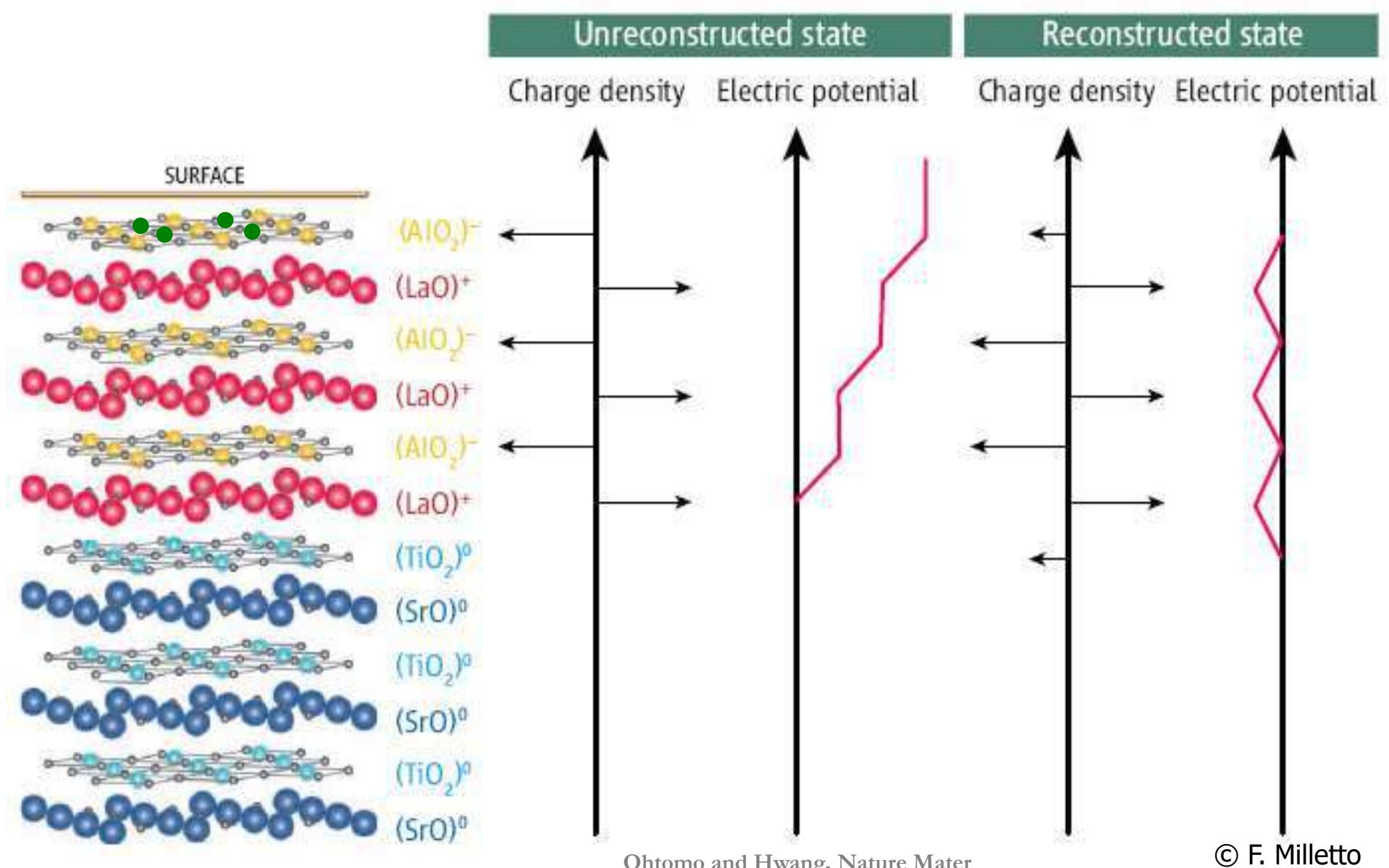
Interfacial conductivity *vs.* LAO thickness



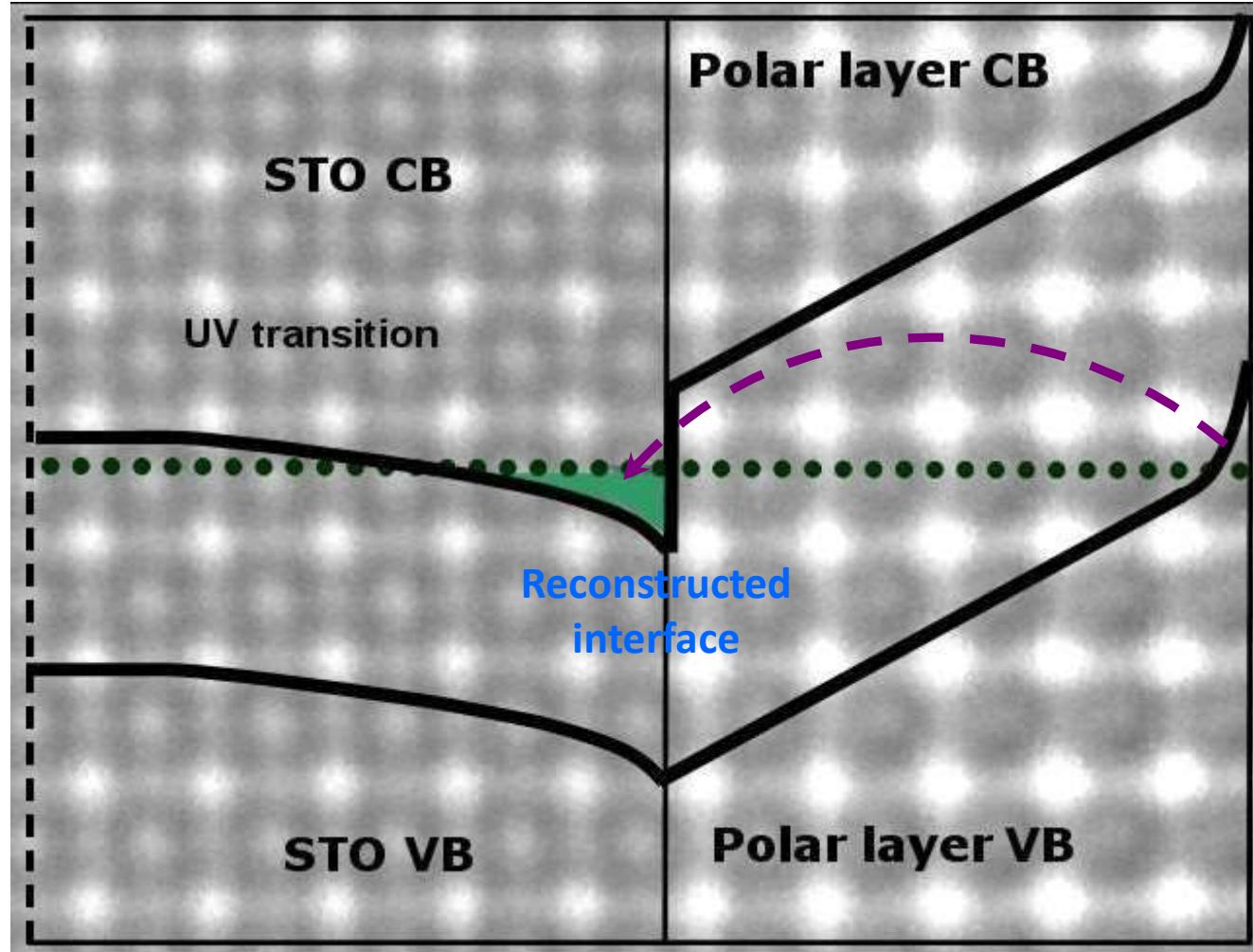
→ **Critical thickness** threshold for conductivity =  
**4 unit cells** (uc) of LAO

S. Thiel et al., Science 313, 1942 (2006)

## Mechanism for 2DEG formation : electronic reconstruction



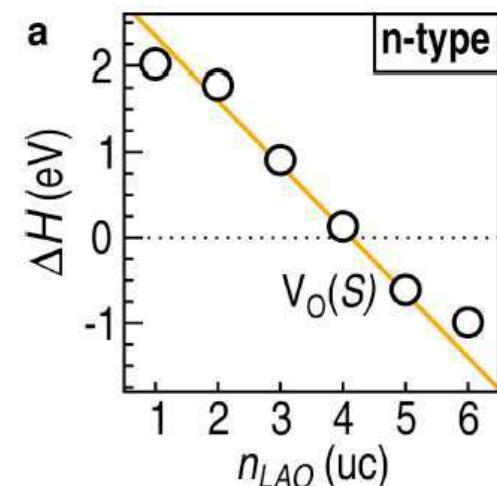
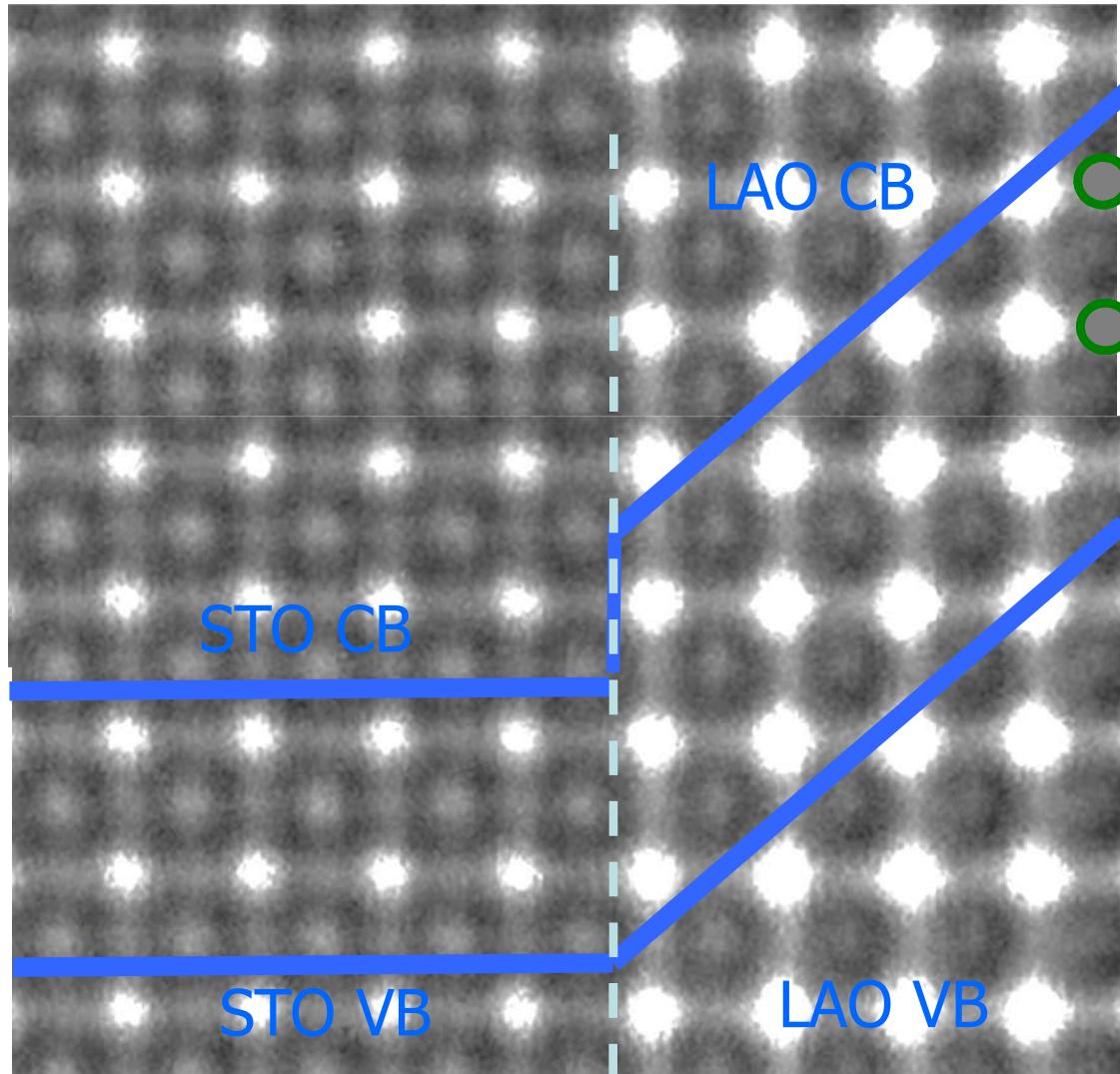
## Mechanism for 2DEG formation : electronic reconstruction



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# Mechanism for 2DEG formation : oxygen vacancies at LAO surface

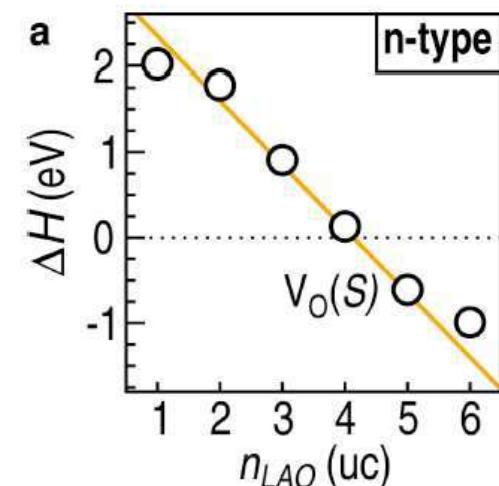
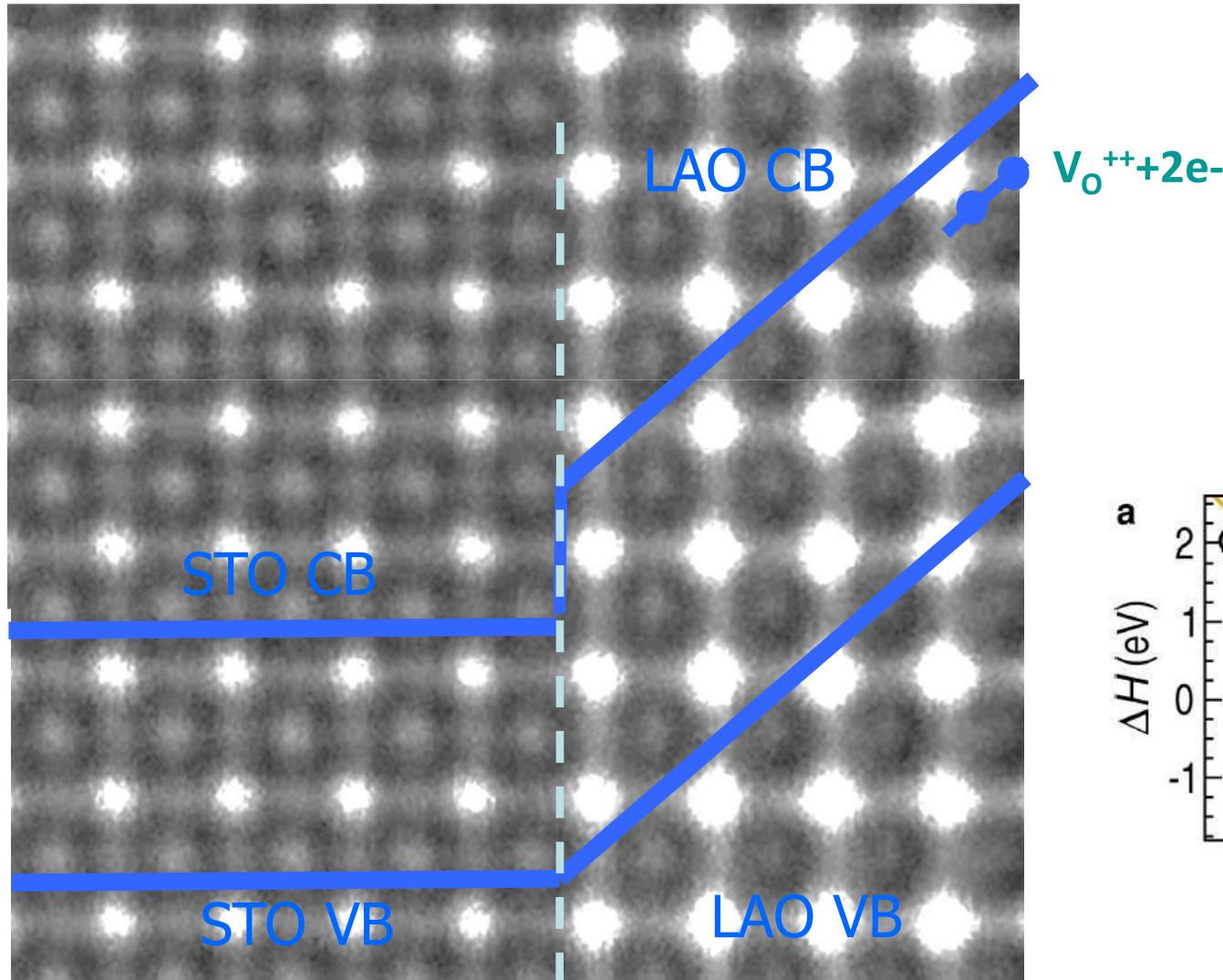
Yu and Zunger, Nature Comm. 5, 5118 (2014)



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# Mechanism for 2DEG formation : oxygen vacancies at LAO surface

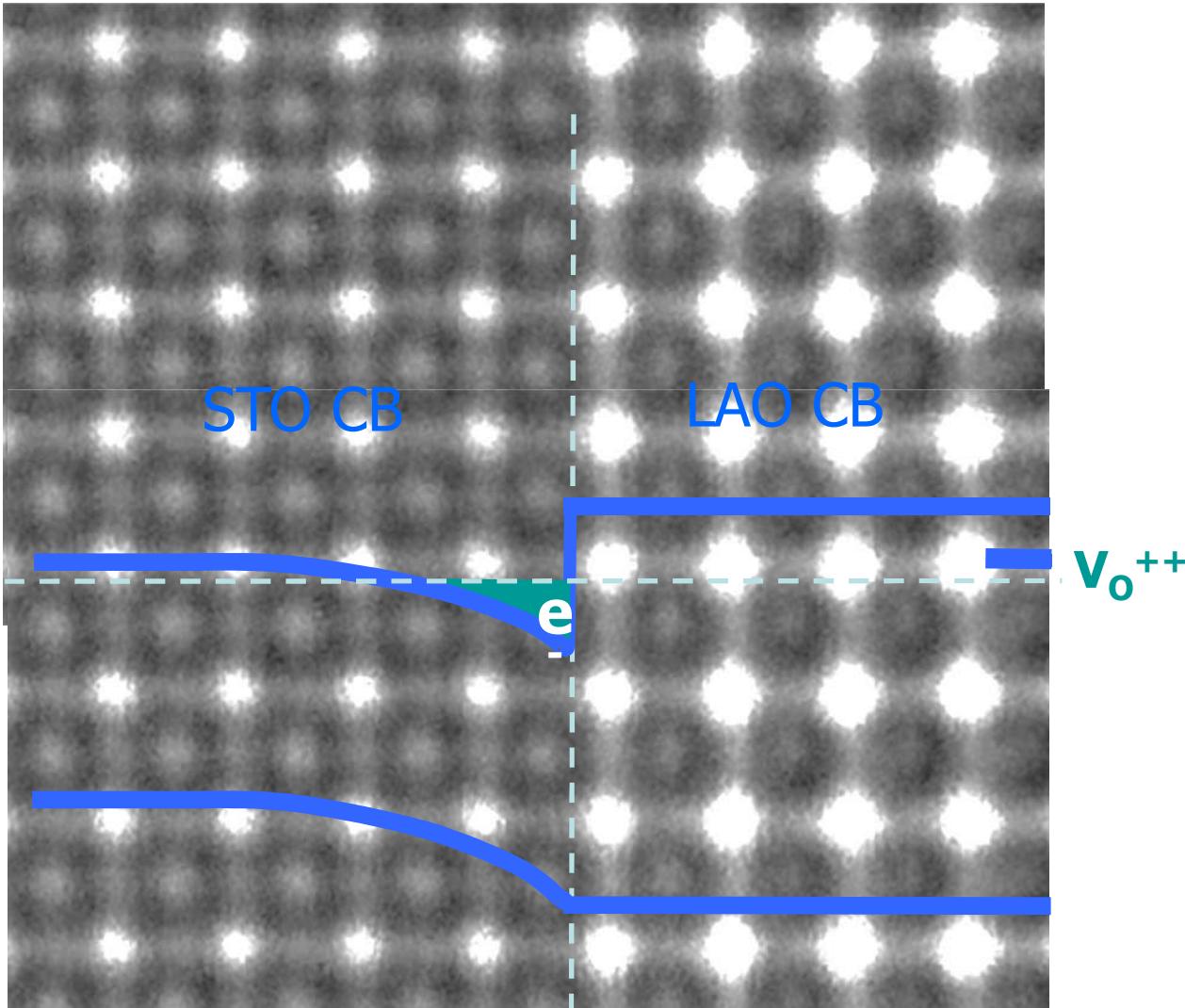
Yu and Zunger, Nature Comm. 5, 5118 (2014)



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Yu and Zunger, Nature Comm. 5, 5118 (2014)



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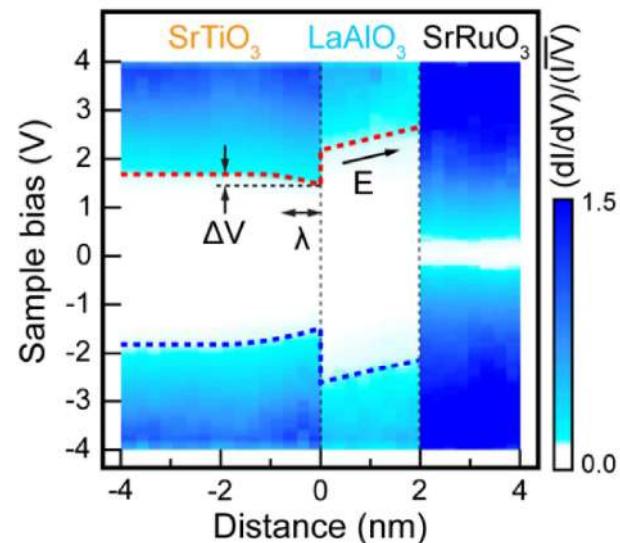
## Mechanism for 2DEG formation : conclusion

### So which mechanism is it ?

Hard to tell because

- Both mechanisms require a polar interface, i.e. no 2DEG for SrO-terminated STO
- Both mechanisms lead to a critical thickness of 4 unit cells
- But after 2DEG formation, there should be an E field in the 2DEG in the polar catastrophe scenario, but not in Yu and Zunger's model

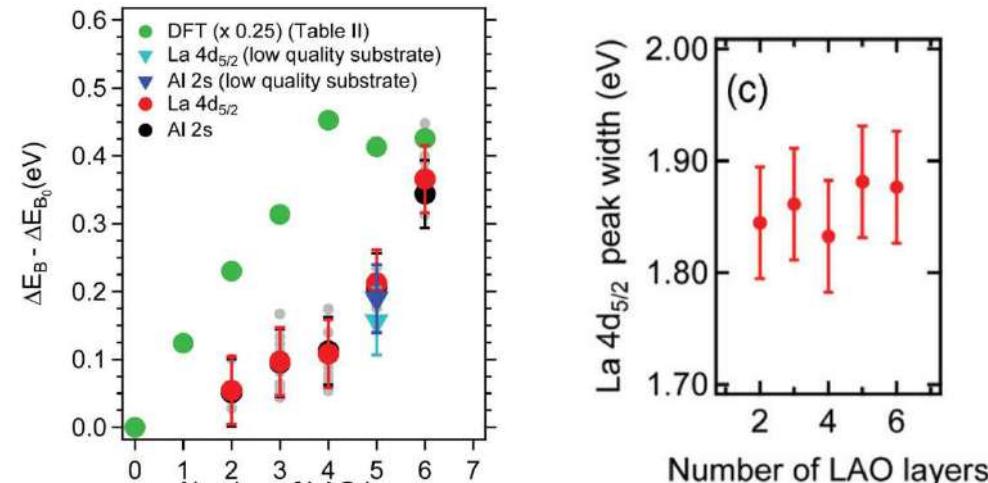
#### Cross section scanning tunnel spectroscopy



- Electric field visible in LAO

Huang et al, PRL 109, 246807 (2012)

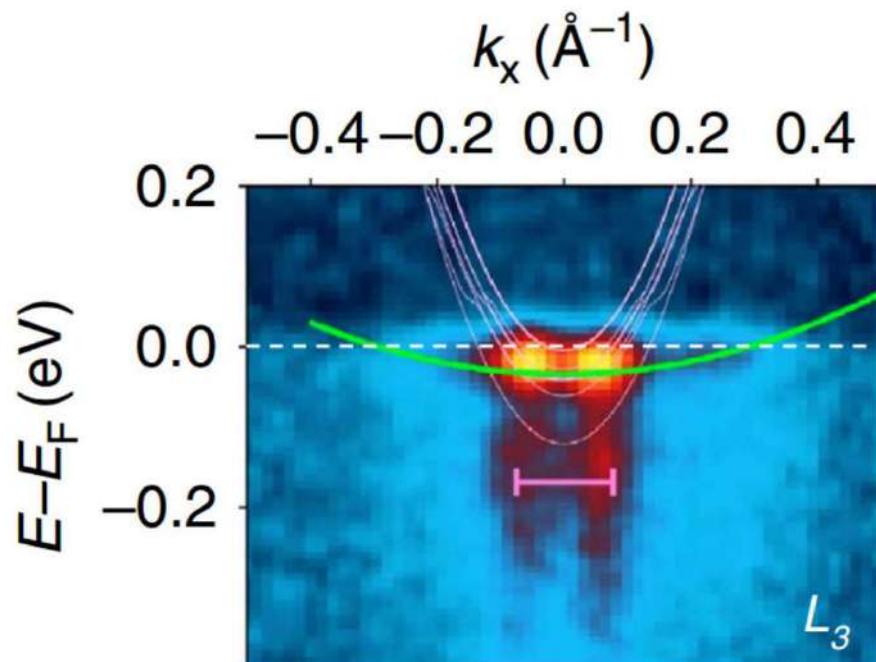
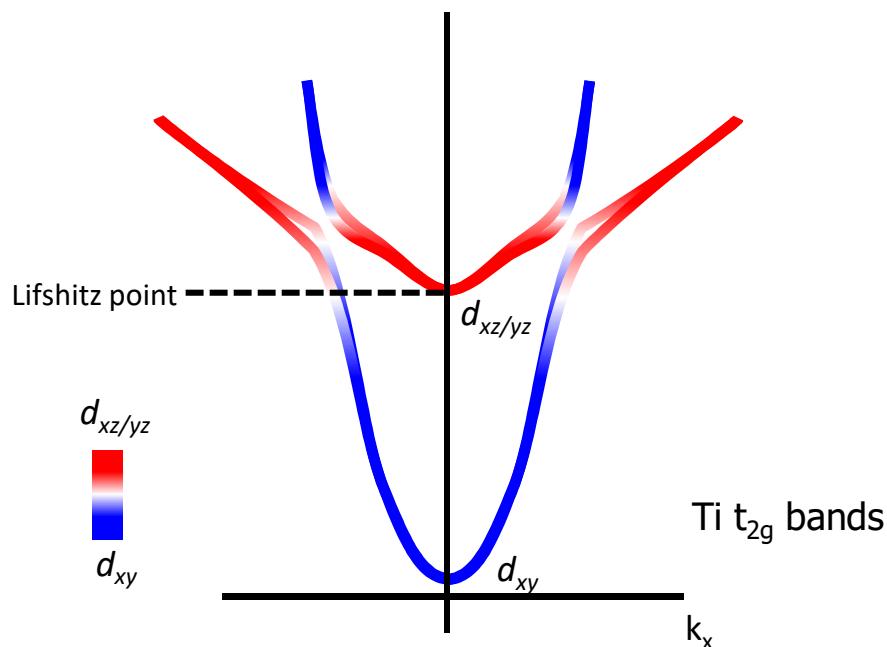
#### Hard X-ray photoemission spectroscopy



- Core level shifts disagree with theory and no broadening

Slooten et al, PRB 87, 085128 (2013)

## Electronic structure of the 2DEG



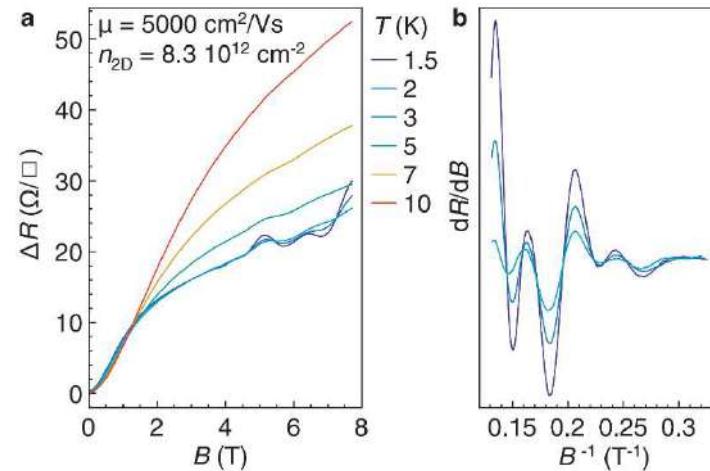
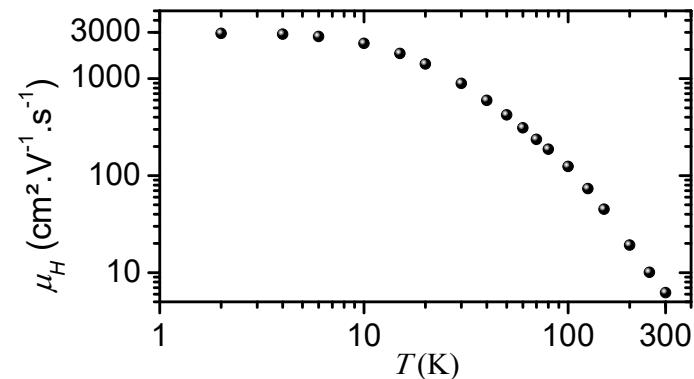
- Compared to bulk STO, the degenerescence of the  $t_{2g}$  states is lifted (splitting is 50-100 meV)
- Low lying  $d_{xy}$  band with light mass
- Above Lifshitz point, onset of second band with  $d_{xz/yz}$  character and heavier mass
- Avoided crossing due to orbital mixing and spin orbit coupling

- Experimental ARPES mapping of the band structure in LAO/STO
- Poor signal quality due to LAO overlayer
- Yet, the main features of the band structure are observed.

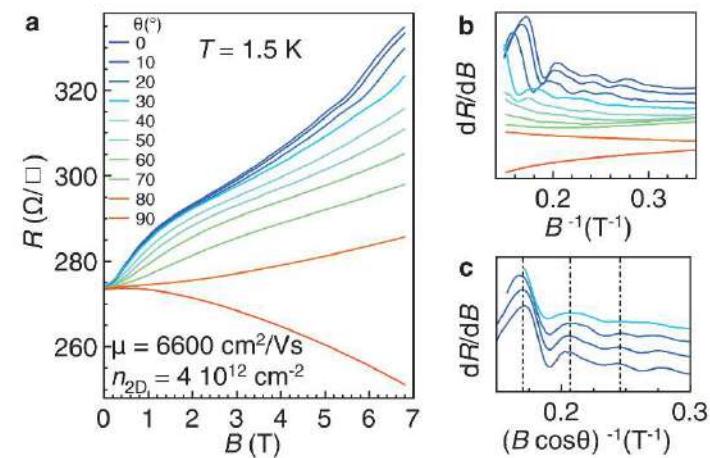
Cancellieri et al, PRB 89, 121412 (2014)

# Electronic properties of the 2DEG

## High electron mobility

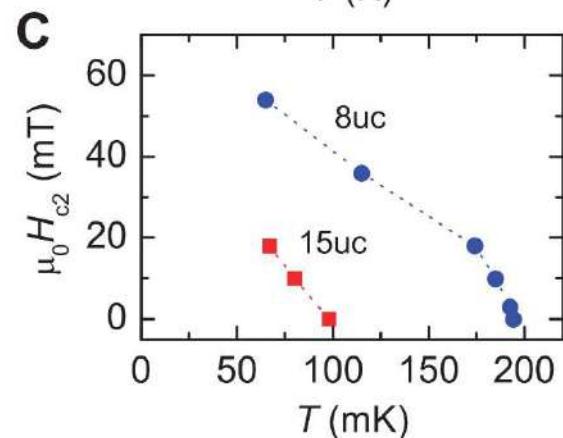
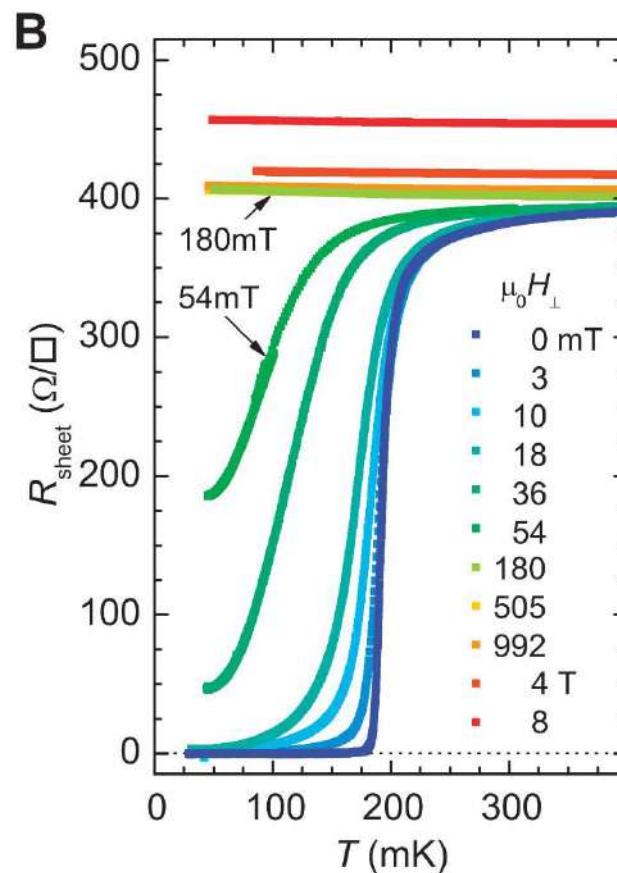
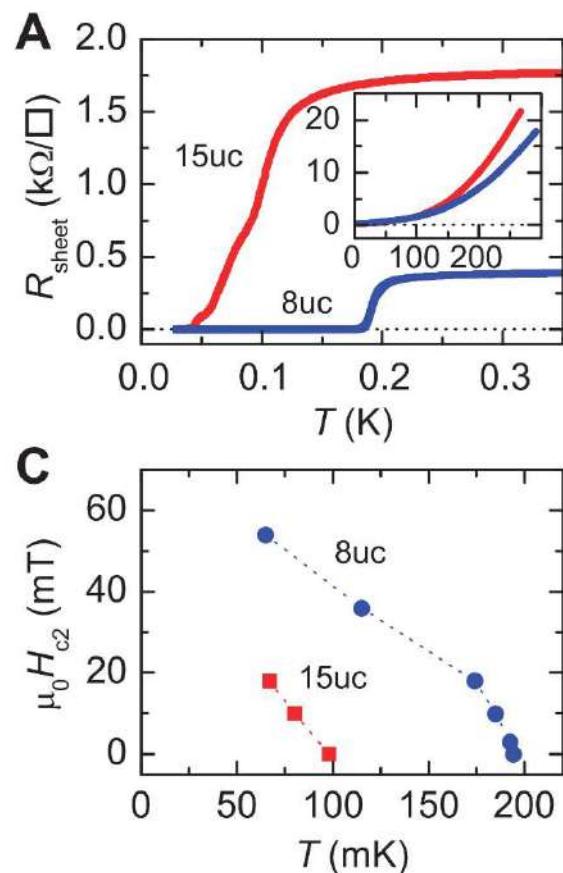


- At low T electron mobility can reach a few 1000 cm<sup>2</sup>/Vs
- Accordingly, Shubnikov de Hass oscillations can be observed (criterion is  $B\mu \gg 1$ )
- Angle dependence of the oscillations reveal the 2D nature of the 2DEG



Caviglia et al, PRL 105, 236802 (2010)

## Superconductivity

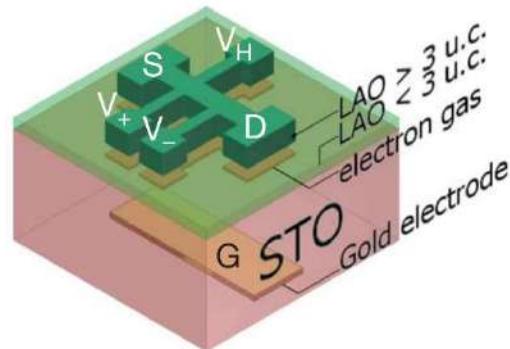


- 2DEG is superconducting at  $T < 200$  mK (like n-type doped bulk STO)
- Superconductivity is 2 dimensional

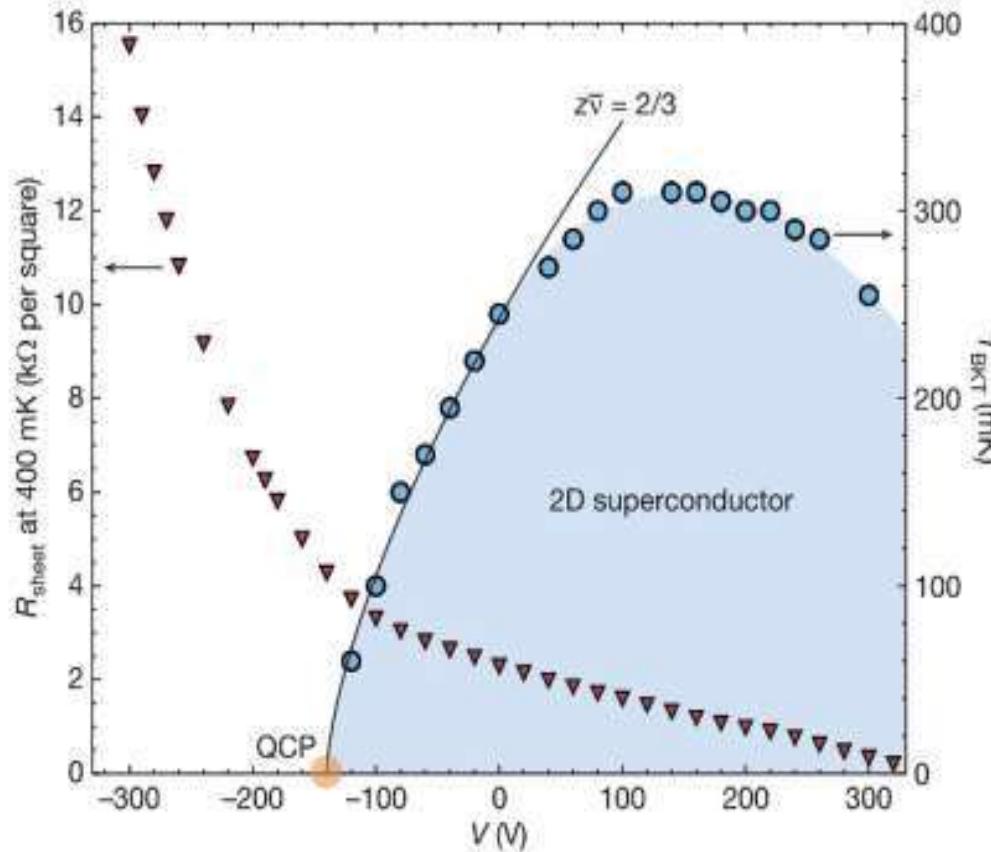
N. Reyren et al., Science 317, 1196 (2007)

# Electronic properties of the 2DEG

## Electrostatic gating

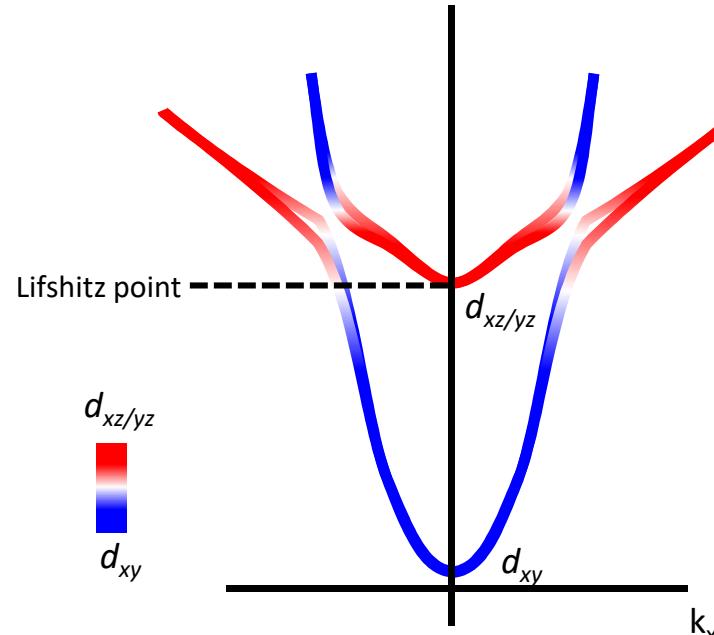
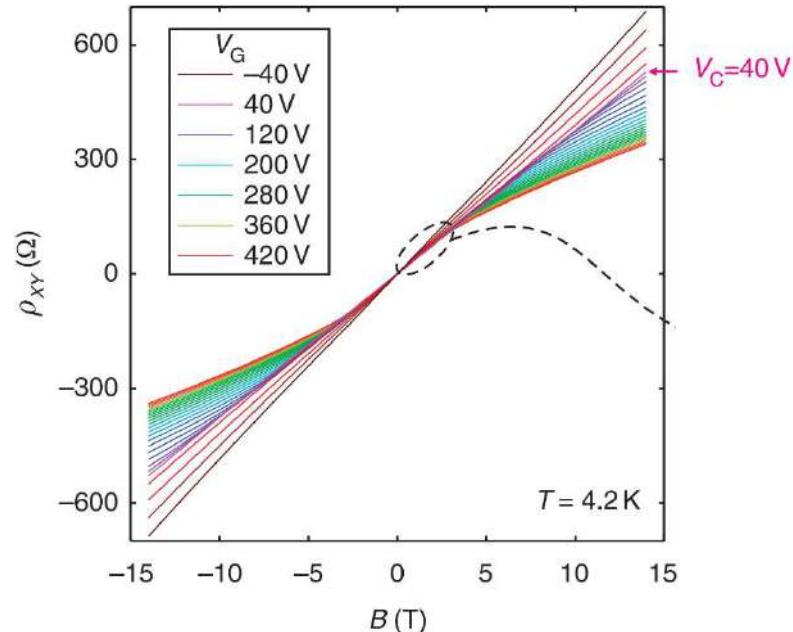


A. D. Caviglia et al.,  
Nature 456, 624 (2008)



- Positive gate voltage accumulates electrons in the 2DEG and negative gate voltage depletes electrons
- Accordingly, the 2DEG normal state resistance varies over more than one order of magnitude
- Superconductivity  $T_c$  is modulated in a dome-like dependence

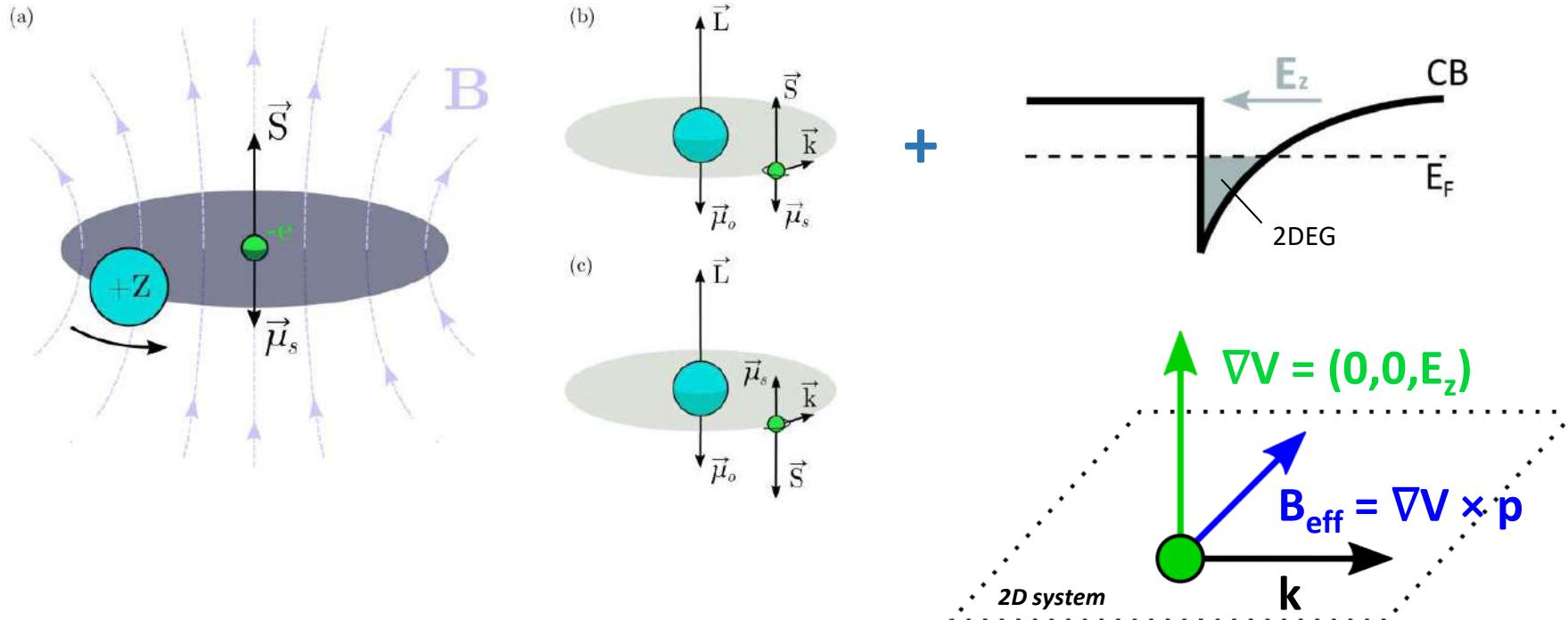
## Electrostatic gating



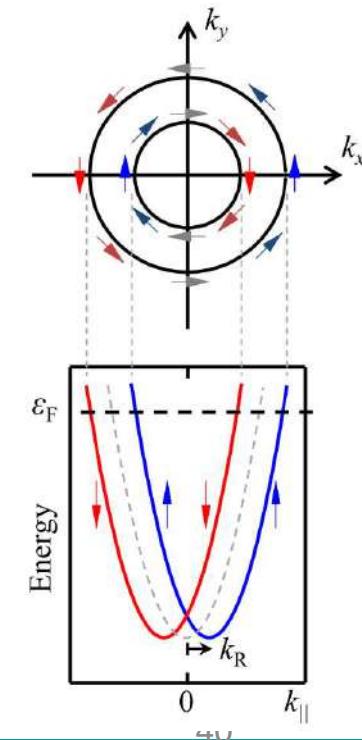
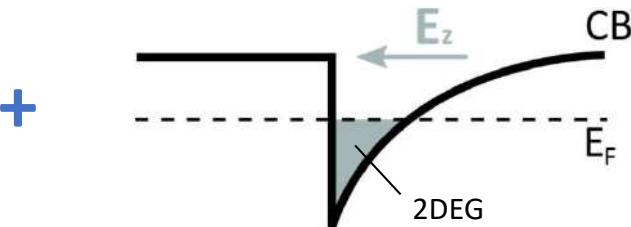
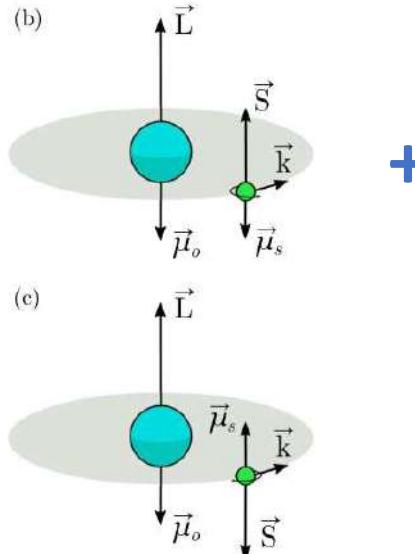
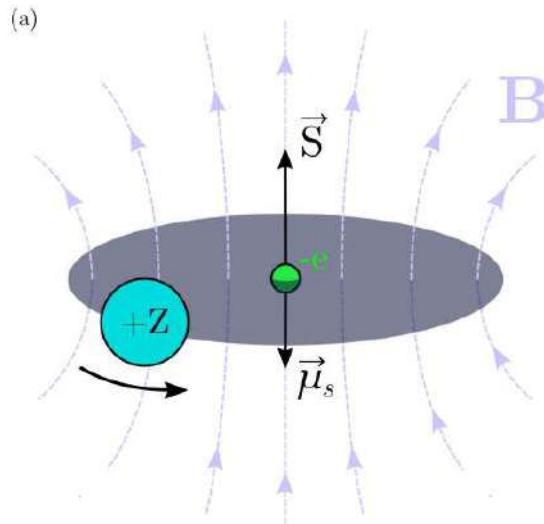
- At low carrier density (negative gate voltages), the Hall effect is linear : one type of carriers
- Upon adding carriers, the Hall effect becomes non-linear : two types of carriers
- Gate induced Lifshitz transition (at  $1.5\text{-}2.5 \cdot 10^{13} \text{ cm}^{-2}$ )

Joshua et al, Nature Comm. 3, 1129 (2012)

## Signatures of spin-orbit coupling



## Signatures of spin-orbit coupling



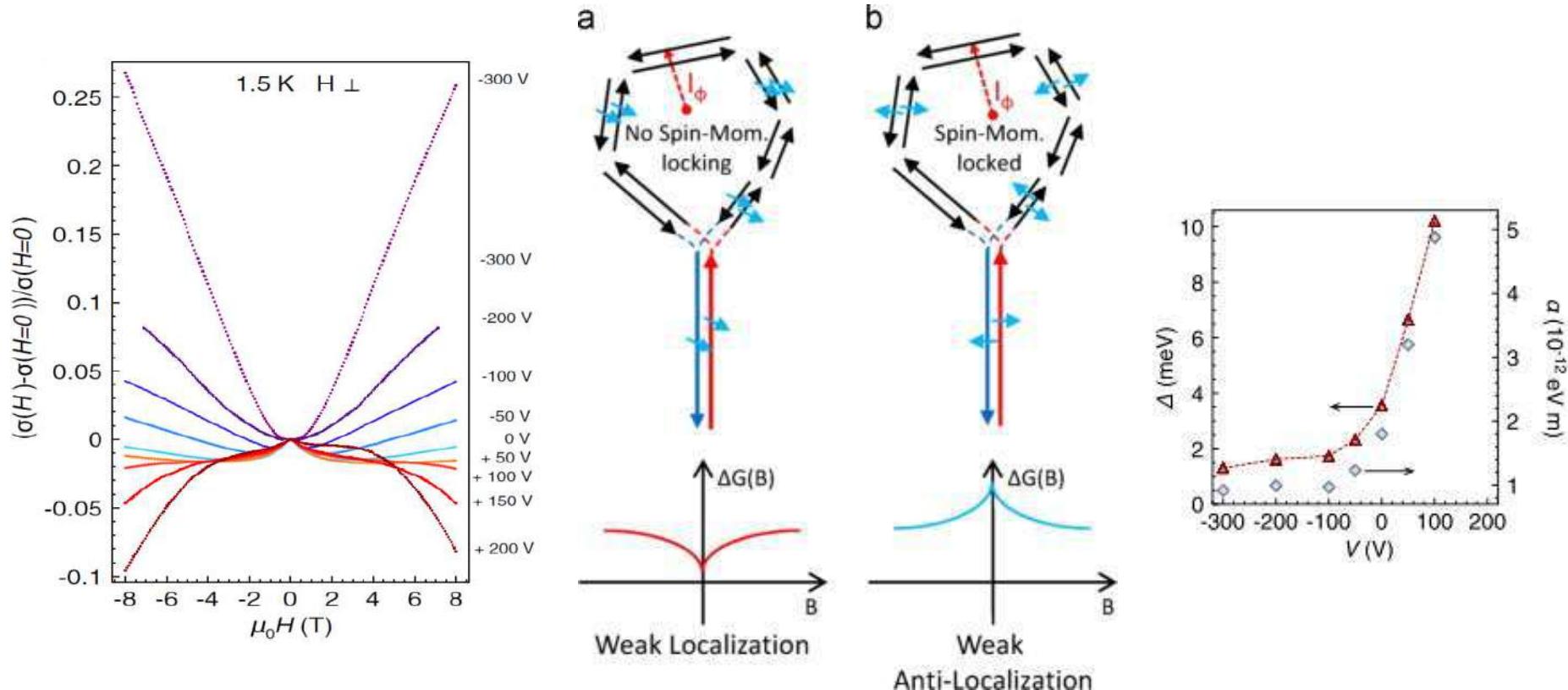
○ Spin orbit interaction + inversion symmetry  
breaking ( $\mathbf{E} \parallel z$ )  
→ **Rashba effect**

$$H_R = \alpha_R (\mathbf{z} \times \mathbf{k}) \cdot \boldsymbol{\sigma}$$

Yu. A. Bychkov & E. I. Rashba,  
JETP Lett. 39, 78 (1984)

$$\varepsilon_k = \frac{\hbar^2 k^2}{2m^*} \pm \alpha_R k$$

## Signatures of spin-orbit coupling



- Magnetoresistance shows signs of weak localization and weak antilocalization
- WAL is a signature of Rashba SOC
- Rashba coefficient  $\alpha_R$  is 10-50 meV. $\text{\AA}$ , i.e. 10-100 times lower than record systems like Bi/Ag interfaces
- Rashba coefficient is strongly tunable by gate voltage

A.D. Caviglia et al., PRL. 104, 126803 (2010)

# 1. **d<sub>0</sub>-oxide-based interfaces**

## 1.1 Physics of bulk SrTiO<sub>3</sub>

## 1.2 LaAlO<sub>3</sub>/SrTiO<sub>3</sub> 2DEGs

## 1.3 Other SrTiO<sub>3</sub> 2DEGs

## 1.4 2DEGs not based on SrTiO<sub>3</sub>

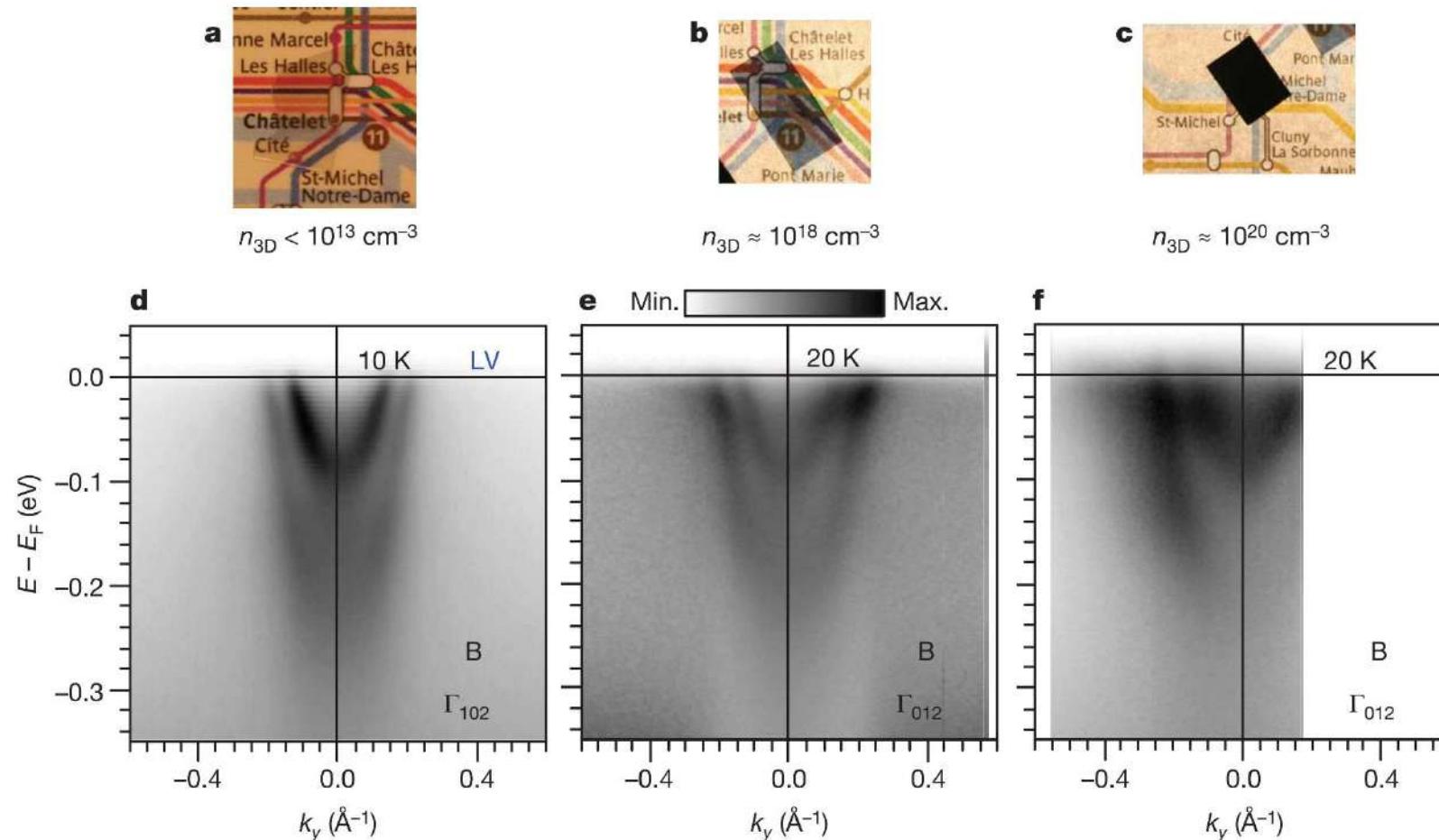
# 2. Interfaces between oxides with partially filled d shells

## 2.1 « Correlated » oxide perovskites

## 2.2 Nickelate/Titanate interfaces

## Other types of SrTiO<sub>3</sub>-based 2DEGs

### A 2DEG at the surface of STO

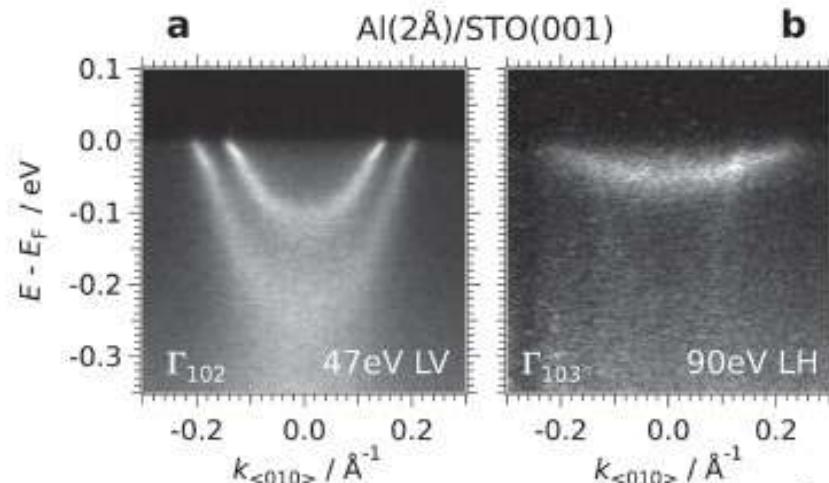
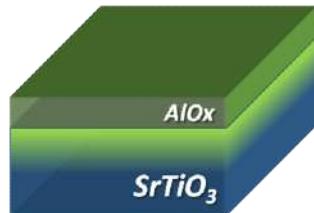


- Fracturing a STO crystal in vacuum creates a 2DEG at its surface
- 2DEG electronic structure very similar to that of the LAO/STO 2DEG

Santander, MB et al, Nature 469, 189 (2011)

## Other types of SrTiO<sub>3</sub>-based 2DEGs

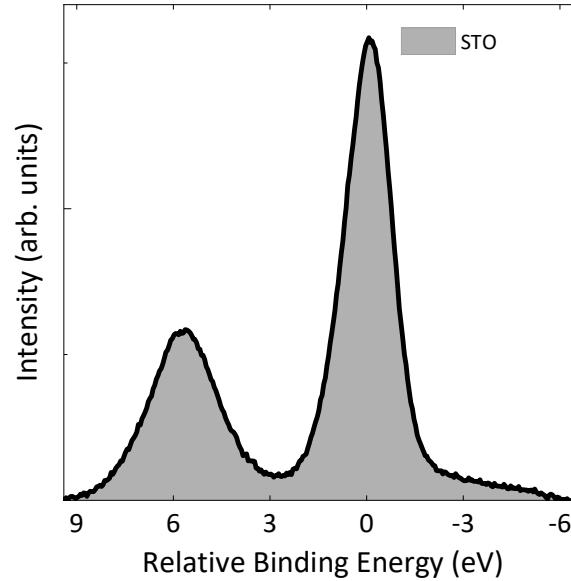
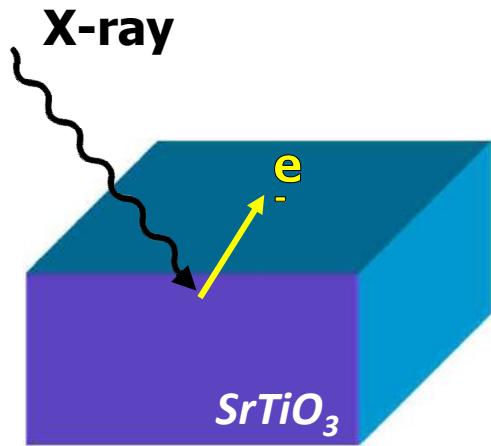
### A 2DEG at Al/STO interfaces



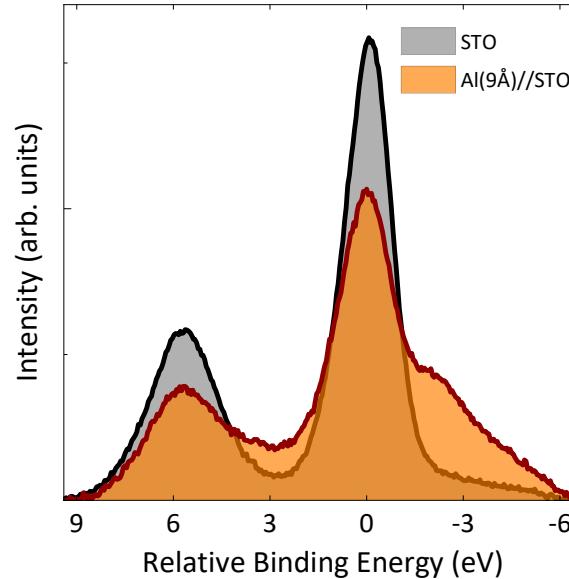
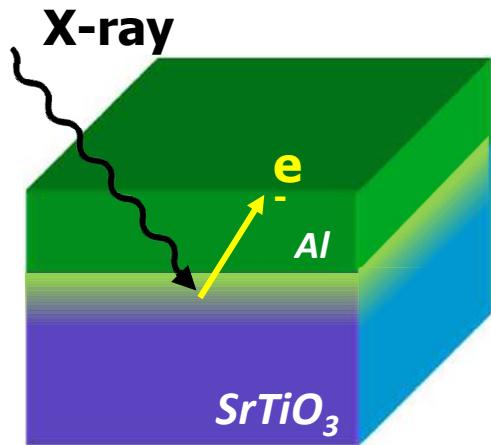
T.C. Rödel et al., Adv. Mater. 28, 1976 (2016)

- Reactive metals → Oxygen vacancies release electrons to form a 2DEG
- Room temperature growth
- What about the polar catastrophe mechanism ? Is the 2DEG formation only driven by oxygen vacancies ?

## A 2DEG in Al/STO

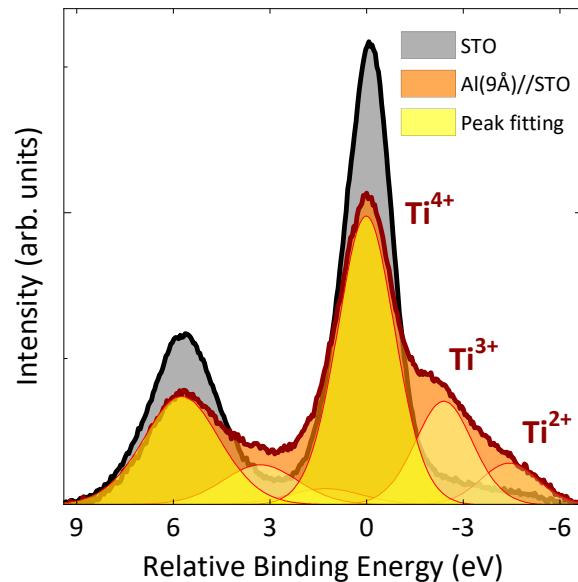
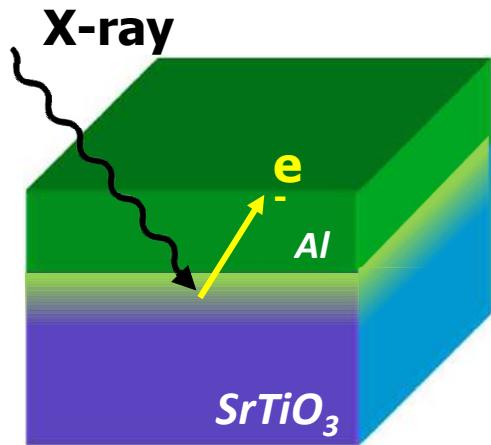


## A 2DEG in Al/STO



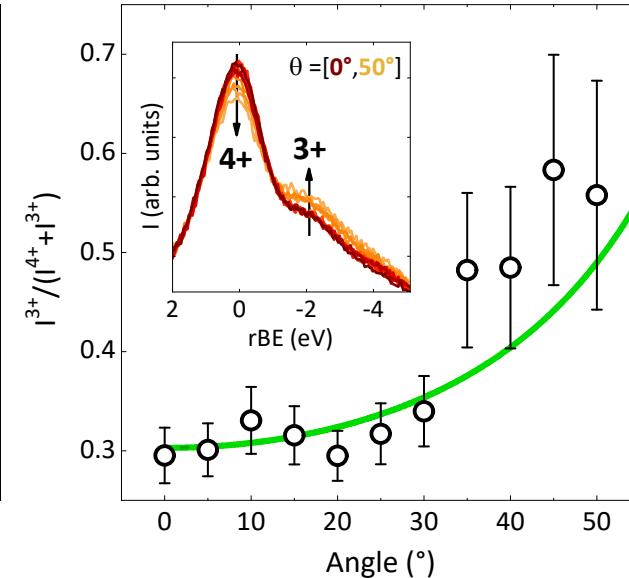
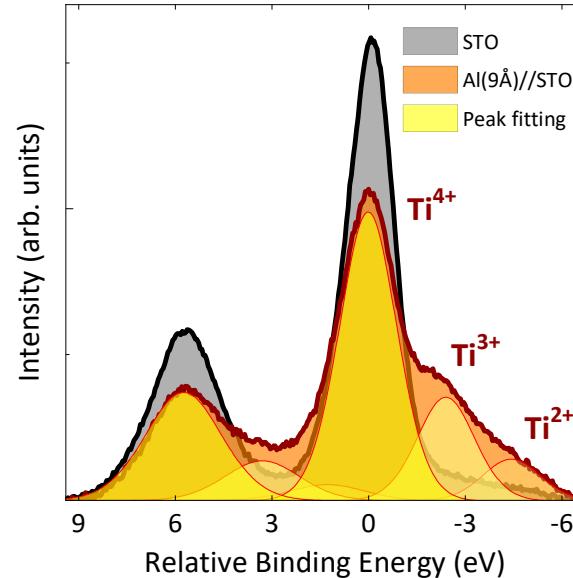
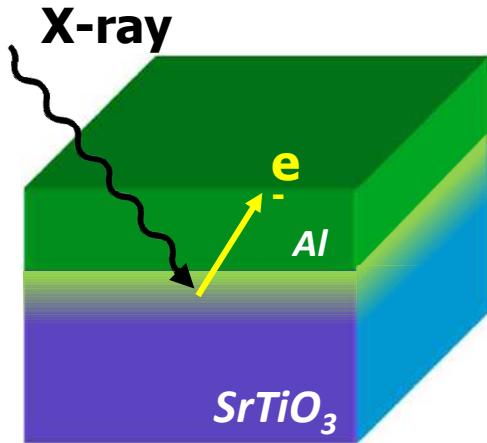
- Deposition of 9 Å of aluminum
- Aluminum pulls oxygen from the STO
- Oxygen vacancies are formed

## A 2DEG in Al/STO



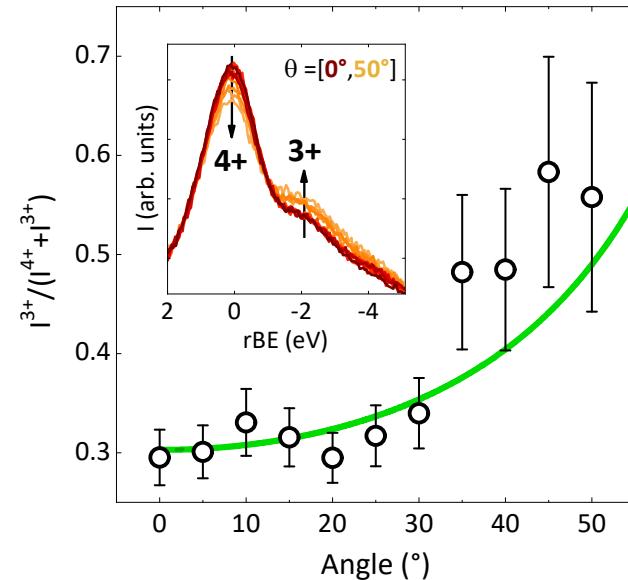
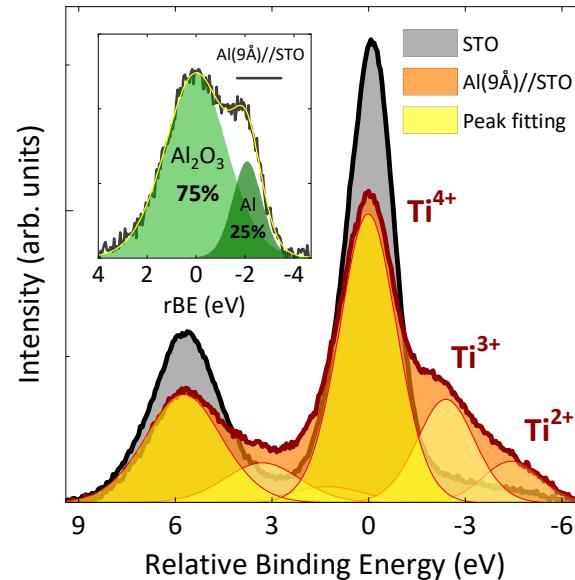
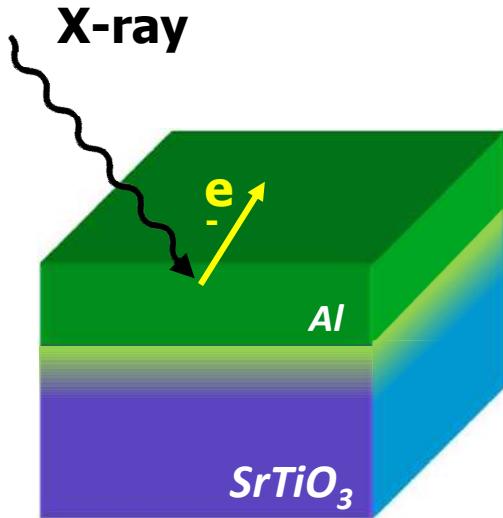
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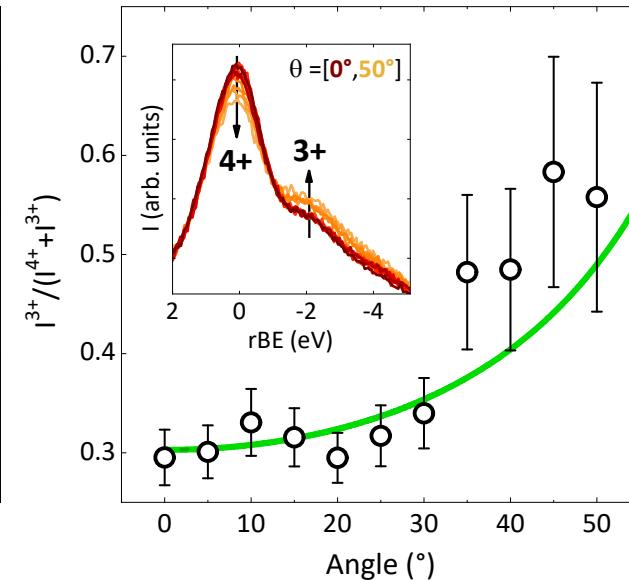
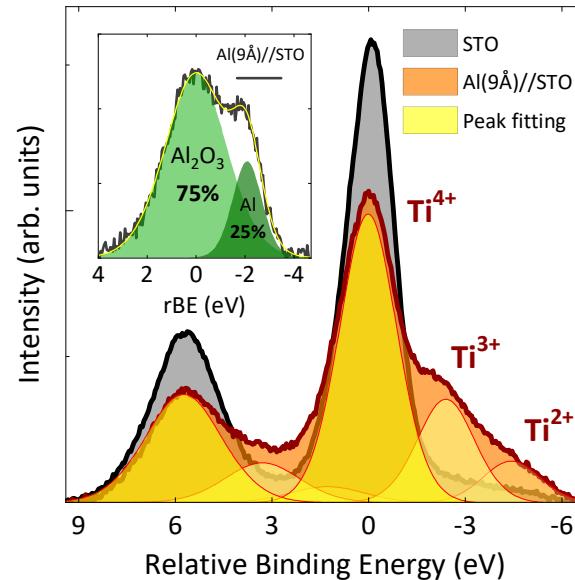
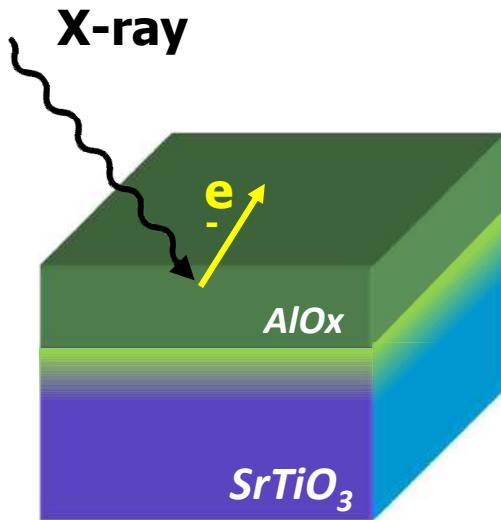
- Deposition of 9 Å of aluminum
- Aluminum pulls oxygen from the STO
- Oxygen vacancies are formed
- A 2DEG emerges at the interface

## A 2DEG in Al/STO



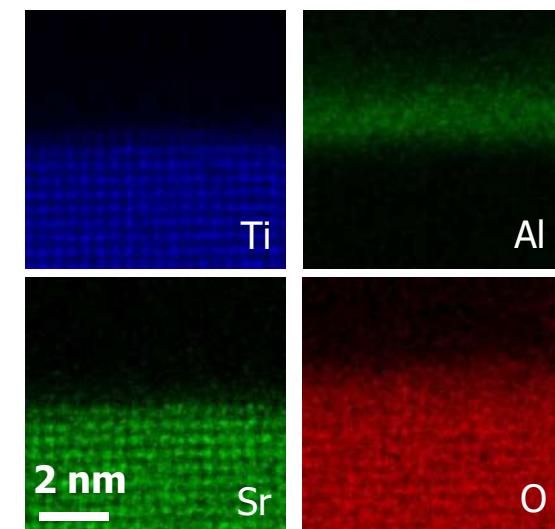
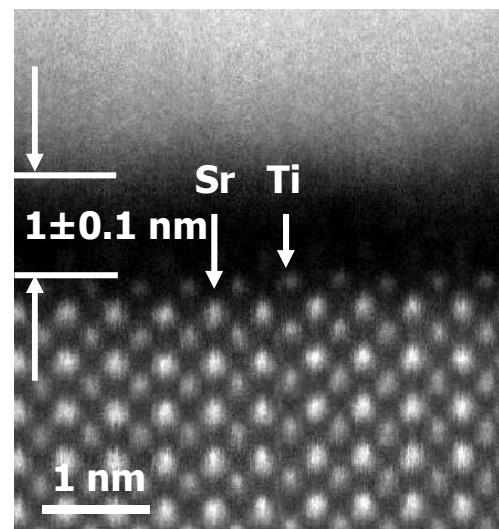
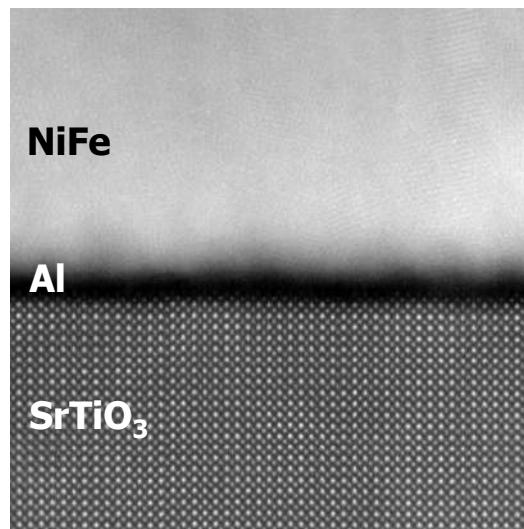
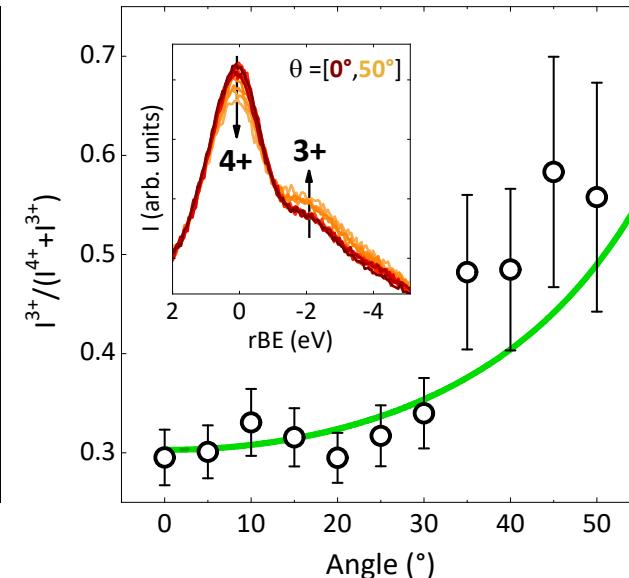
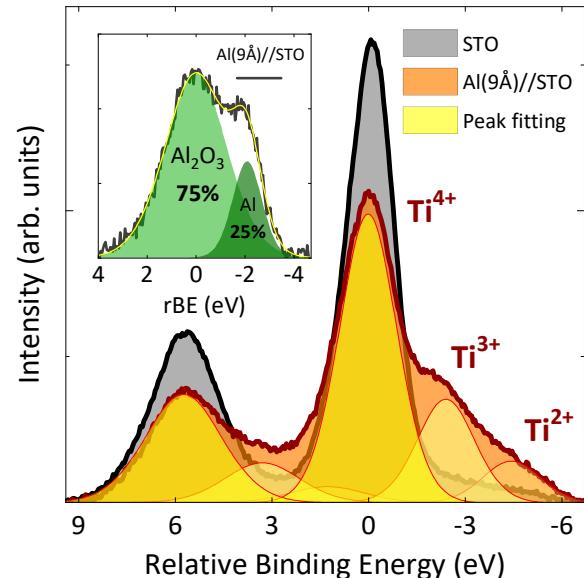
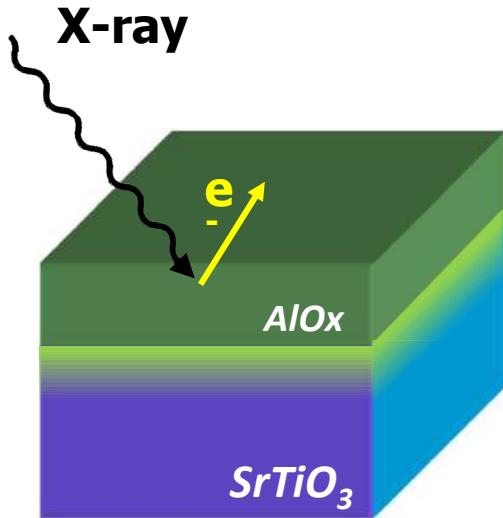
- Deposition of 9 Å of aluminum
- Aluminum pulls oxygen from the STO
- Oxygen vacancies are formed
- A 2DEG emerges at the interface

## A 2DEG in Al/STO



- ➊ Deposition of 9 Å of aluminum
- ➋ Aluminum pulls oxygen from the STO
- ➌ Oxygen vacancies are formed
- ➍ A 2DEG emerges at the interface
- ➎ The deposited aluminum layer is completely oxidized

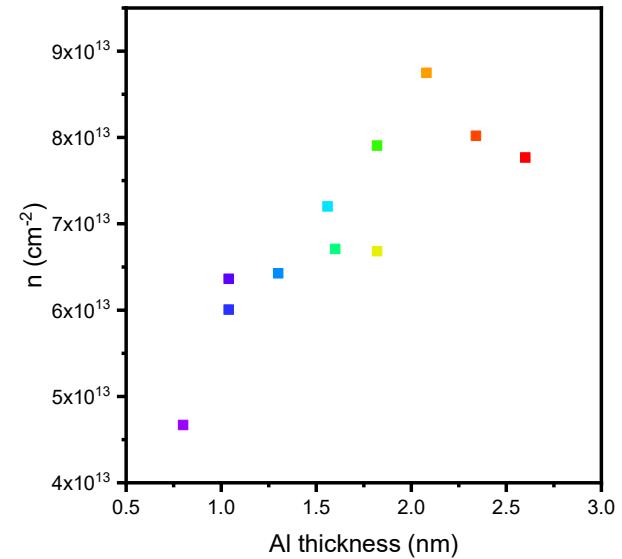
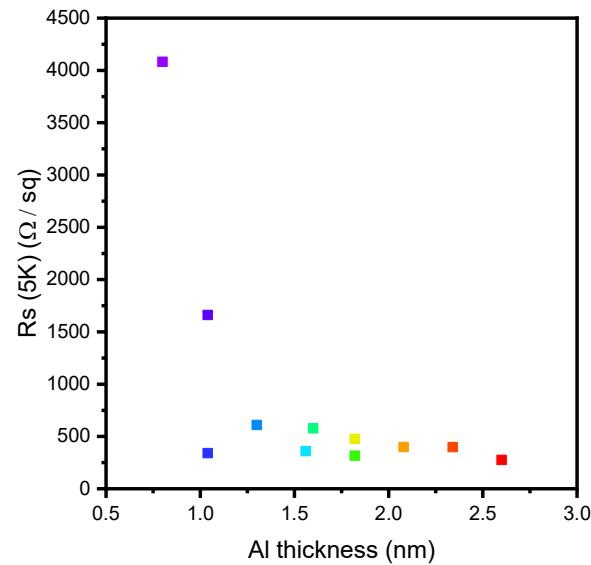
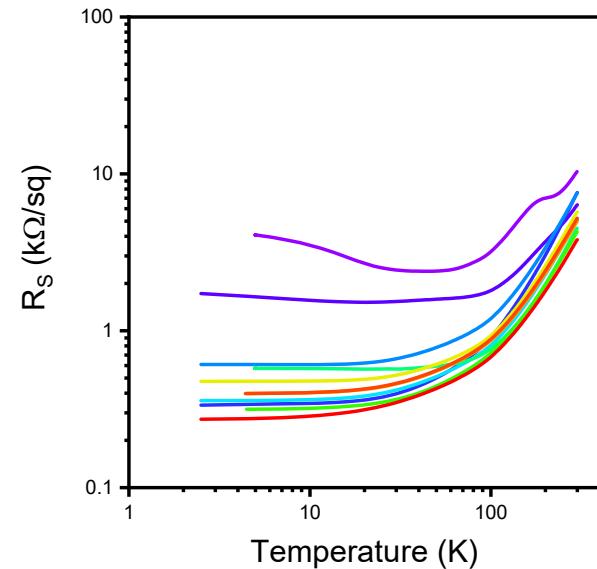
## A 2DEG in Al/STO



Presence of 1 nm AlOx layer at interface with STO

Very little interdiffusion if any

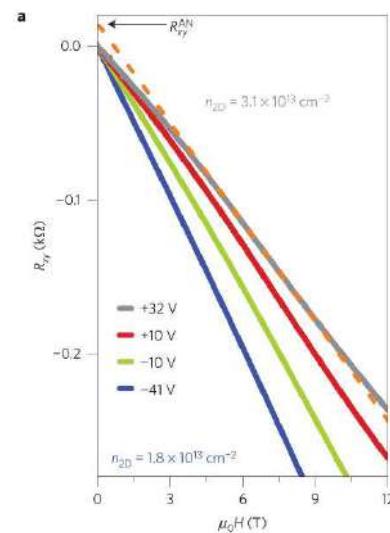
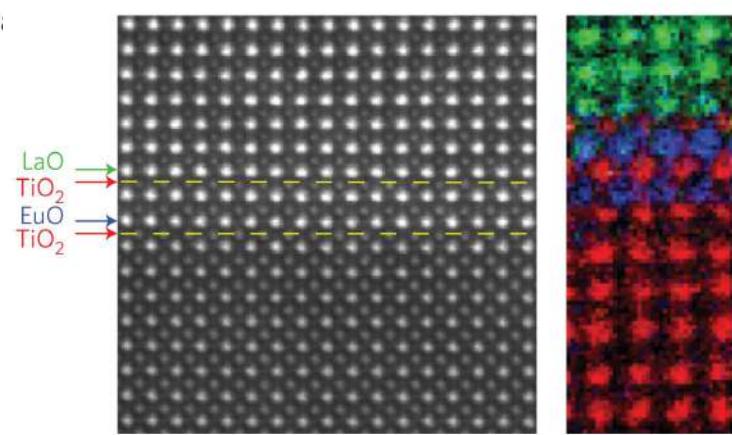
## Dependence of transport properties with Al thickness



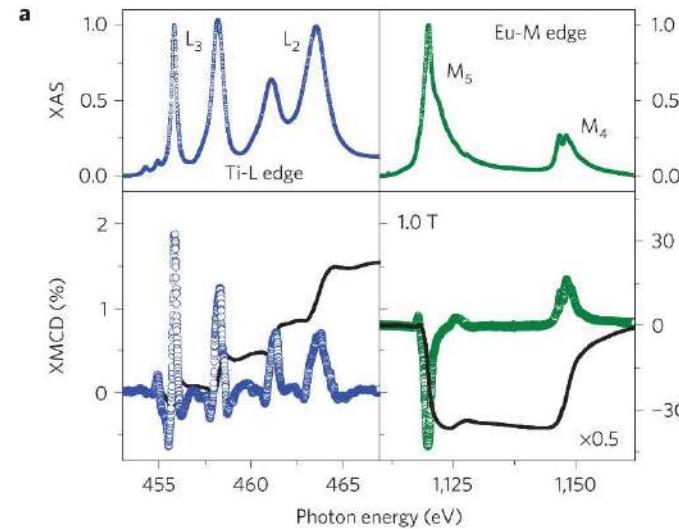
- Conductivity and carrier density increase with Al thickness
- More oxygen vacancies, more carriers
- **Check Luis Moreno's poster tomorrow !**

# Engineering a magnetic 2DEG

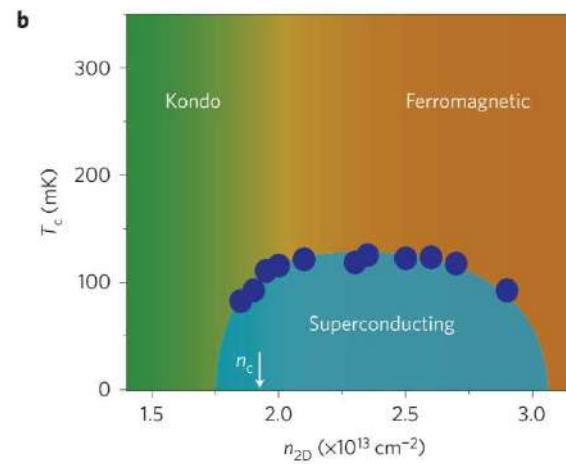
## Insert a monolayer of $\text{EuTiO}_3$ between LAO and STO



○ Anomalous Hall effect

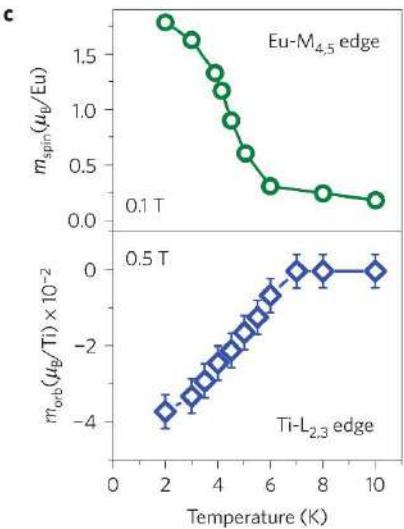


○ XMCD at Ti and Eu edges



○ Coexistence of ferromagnetism and superconductivity, tunable by gate voltage

Stornaiuolo et al, Nature Mater 15, 278 (2016)



### 1. **d<sub>0</sub>-oxide-based interfaces**

#### 1.1 Physics of bulk SrTiO<sub>3</sub>

#### 1.2 LaAlO<sub>3</sub>/SrTiO<sub>3</sub> 2DEGs

#### 1.3 Other SrTiO<sub>3</sub> 2DEGs

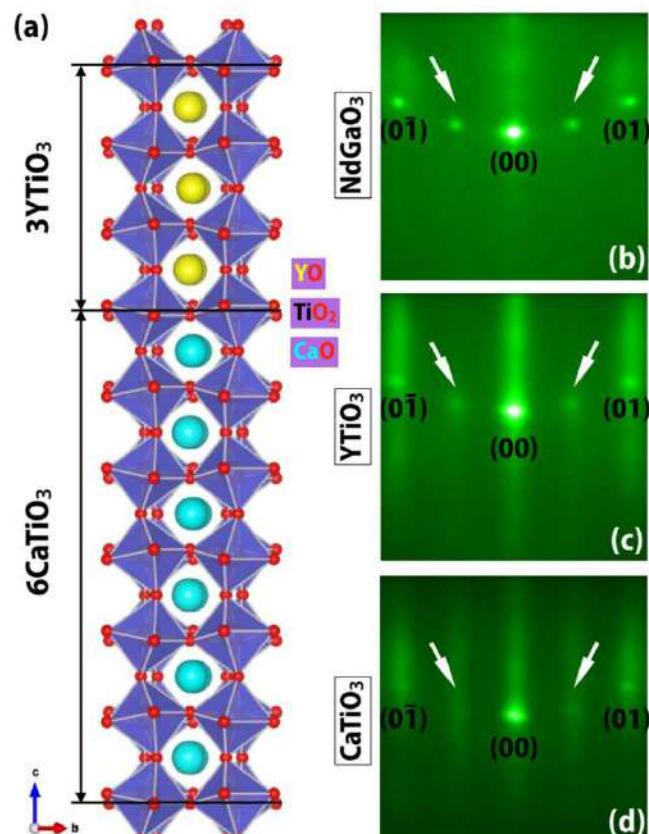
#### 1.4 2DEGs not based on SrTiO<sub>3</sub>

### 2. Interfaces between oxides with partially filled d shells

#### 2.1 « Correlated » oxide perovskites

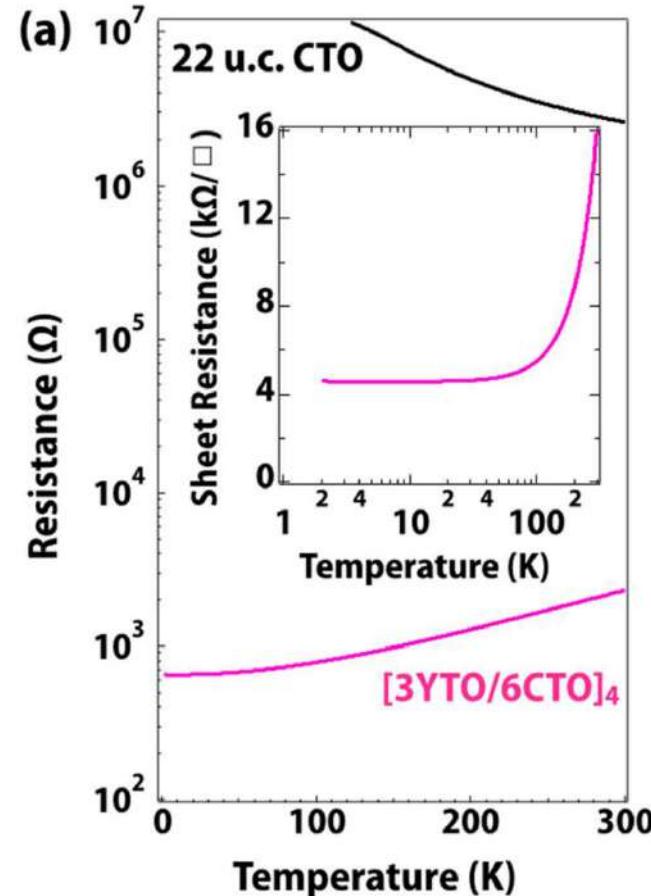
#### 2.2 Nickelate/Titanate interfaces

### Metallic interface between $\text{CaTiO}_3$ and $\text{YTiO}_3$



Liu et al, APL 107 191602 (2015)

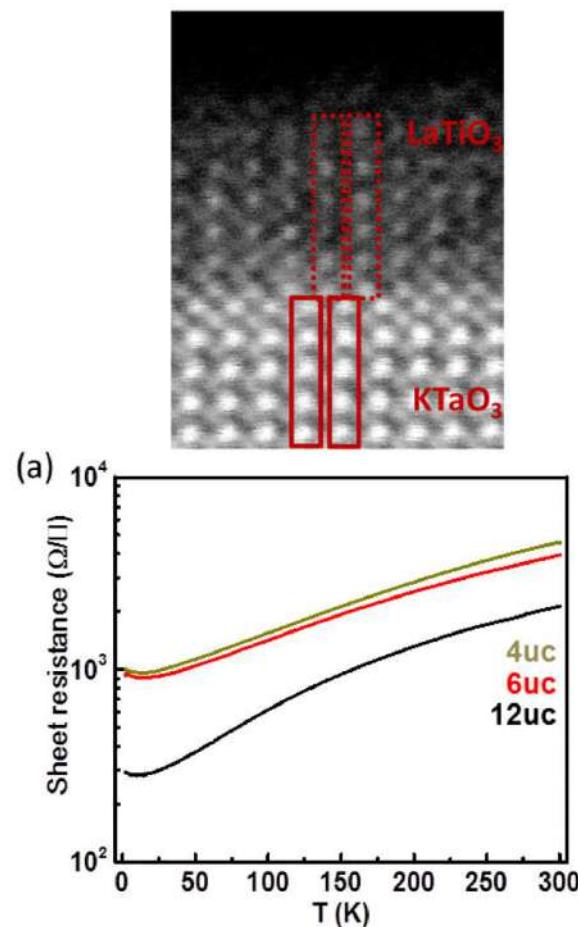
- $\text{CaTiO}_3/\text{YTiO}_3$  bilayer is grown on  $\text{NdGaO}_3$  substrate
- Conductive interface appears (2DEG ?) : charge transfer from YTO to CTO ?



## 2DEGs in non STO systems

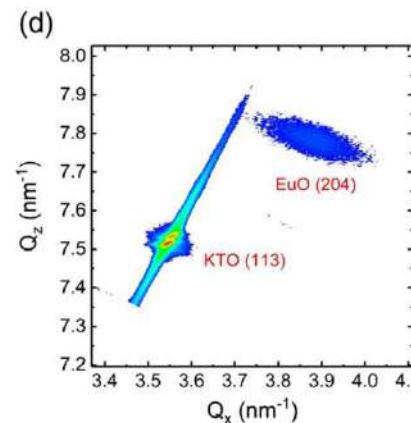
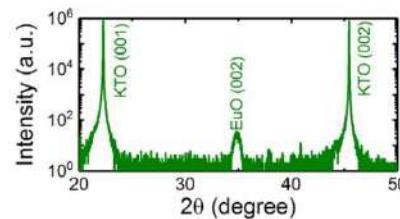
### 2DEGs based on $\text{KTaO}_3$

$\text{LaTiO}_3/\text{KTaO}_3$



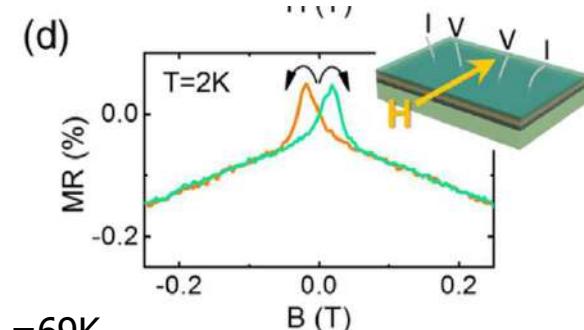
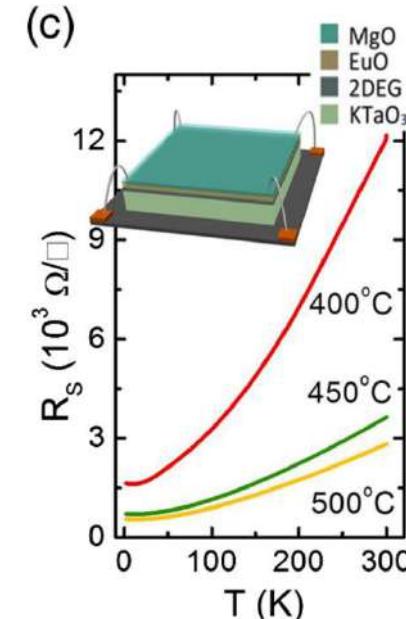
Zou et al, APL Mater. 3, 036104 (2015)

- $\text{LaTiO}_3$  film grown on  $\text{KTaO}_3$  single crystal
- Conductive interface looking like 2DEG



Zhang et al PRL 121, 116803 (2018)

$\text{EuO}/\text{KTaO}_3$



- $\text{EuO}$  is FM insulator with  $T_C=69\text{K}$
- $\text{EuO}$  film grown on  $\text{KTaO}_3$  single crystal
- Conductive interface with magnetoresistance : magnetic 2DEG

### 1. $d_0$ -oxide-based interfaces

#### 1.1 Physics of bulk SrTiO<sub>3</sub>

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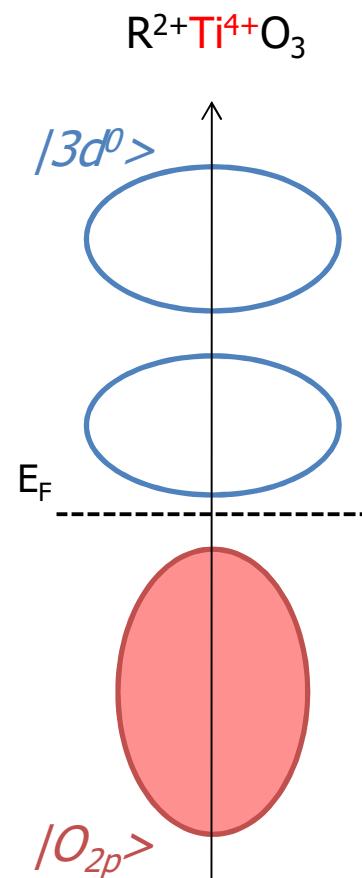
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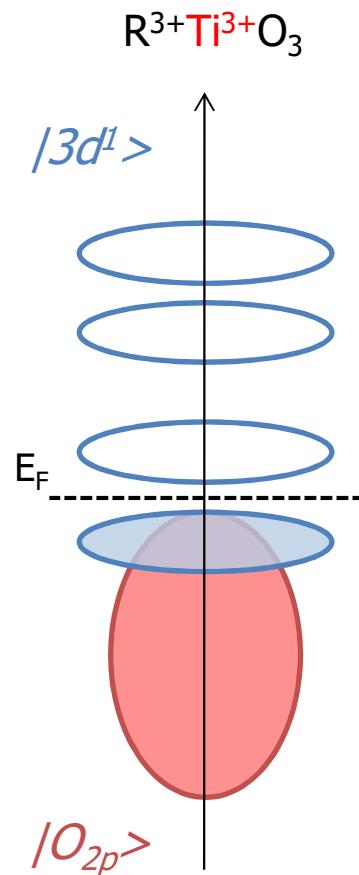
#### 2.2 Nickelate/Titanate interfaces

# Interfaces between oxides with partially filled d shells



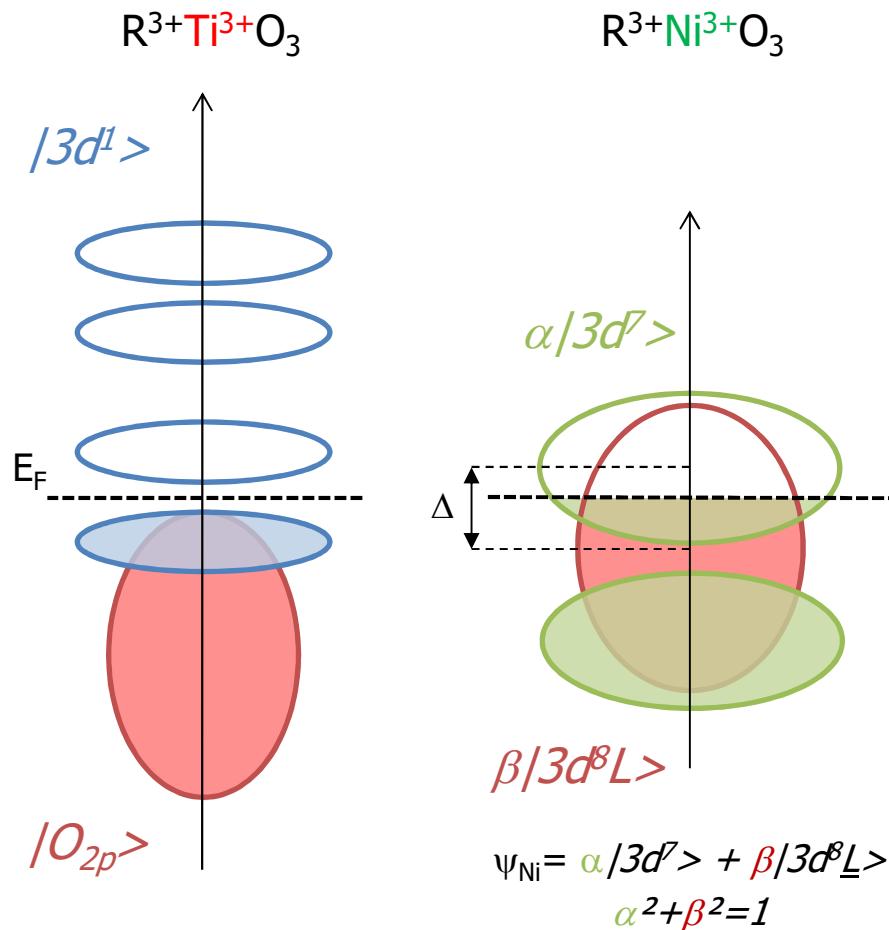
- No **d** electrons
- No magnetism

# Interfaces between oxides with partially filled d shells



- **d** electrons
- Magn/orb order
- **Mott insulator**

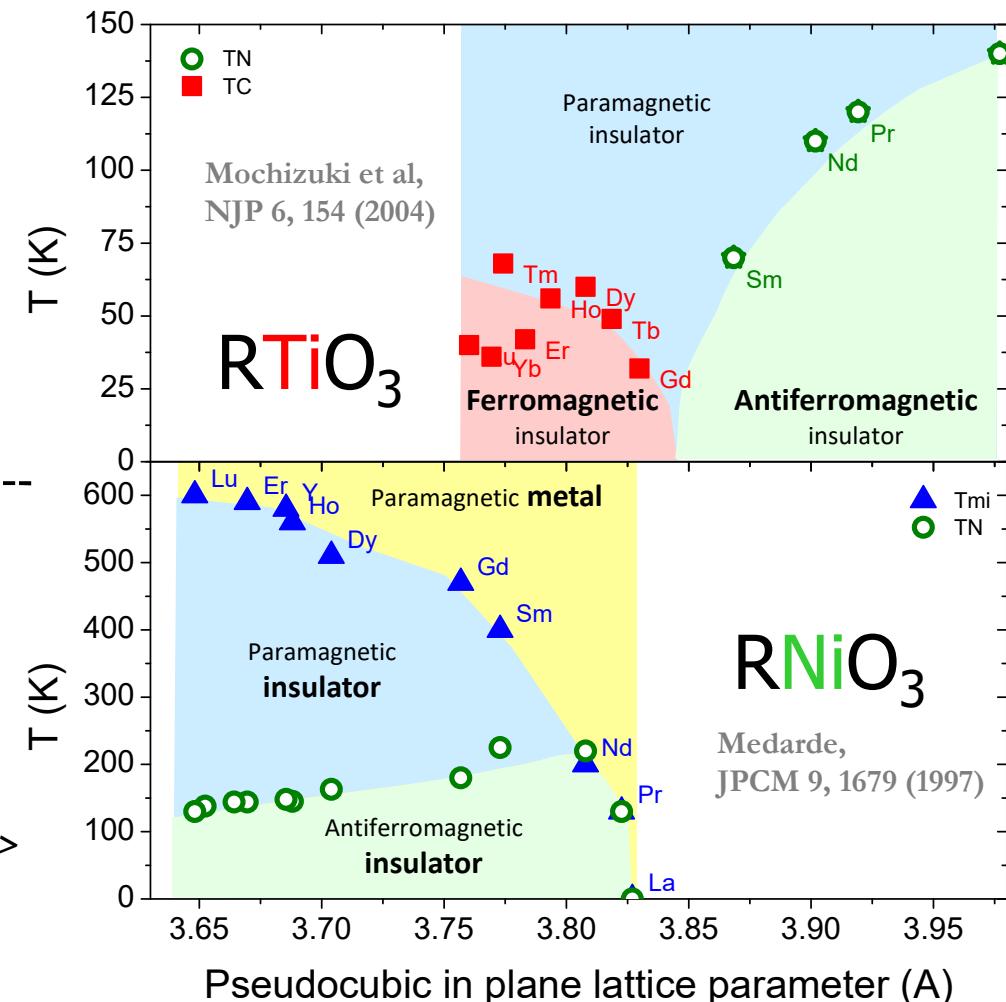
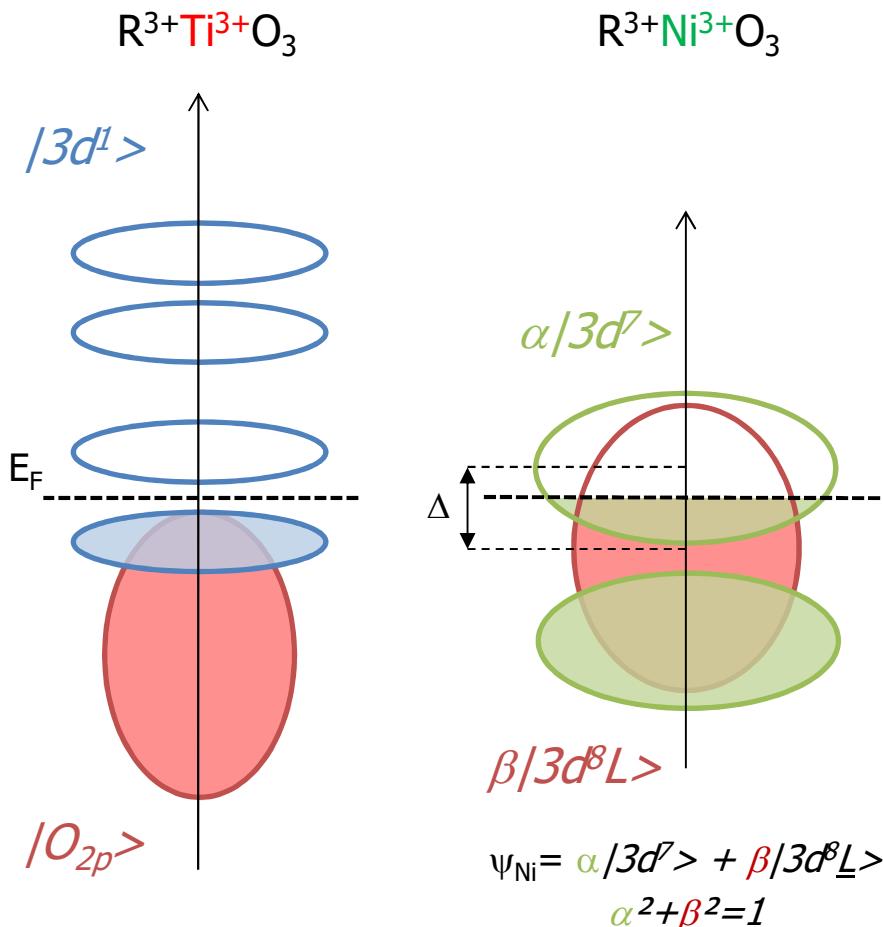
# Interfaces between oxides with partially filled d shells



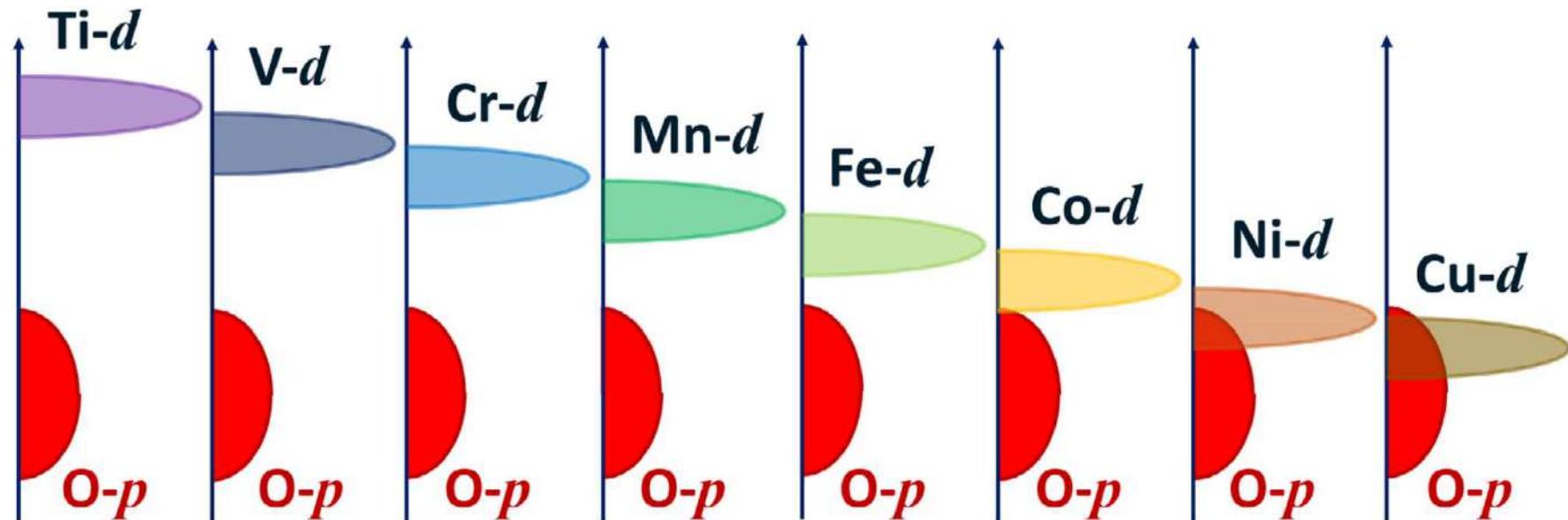
- **d** electrons
- Magn/orb order
- **Mott insulator**

- **d** electrons
- Magn/orb order
- Strong covalence

# Interfaces between oxides with partially filled d shells



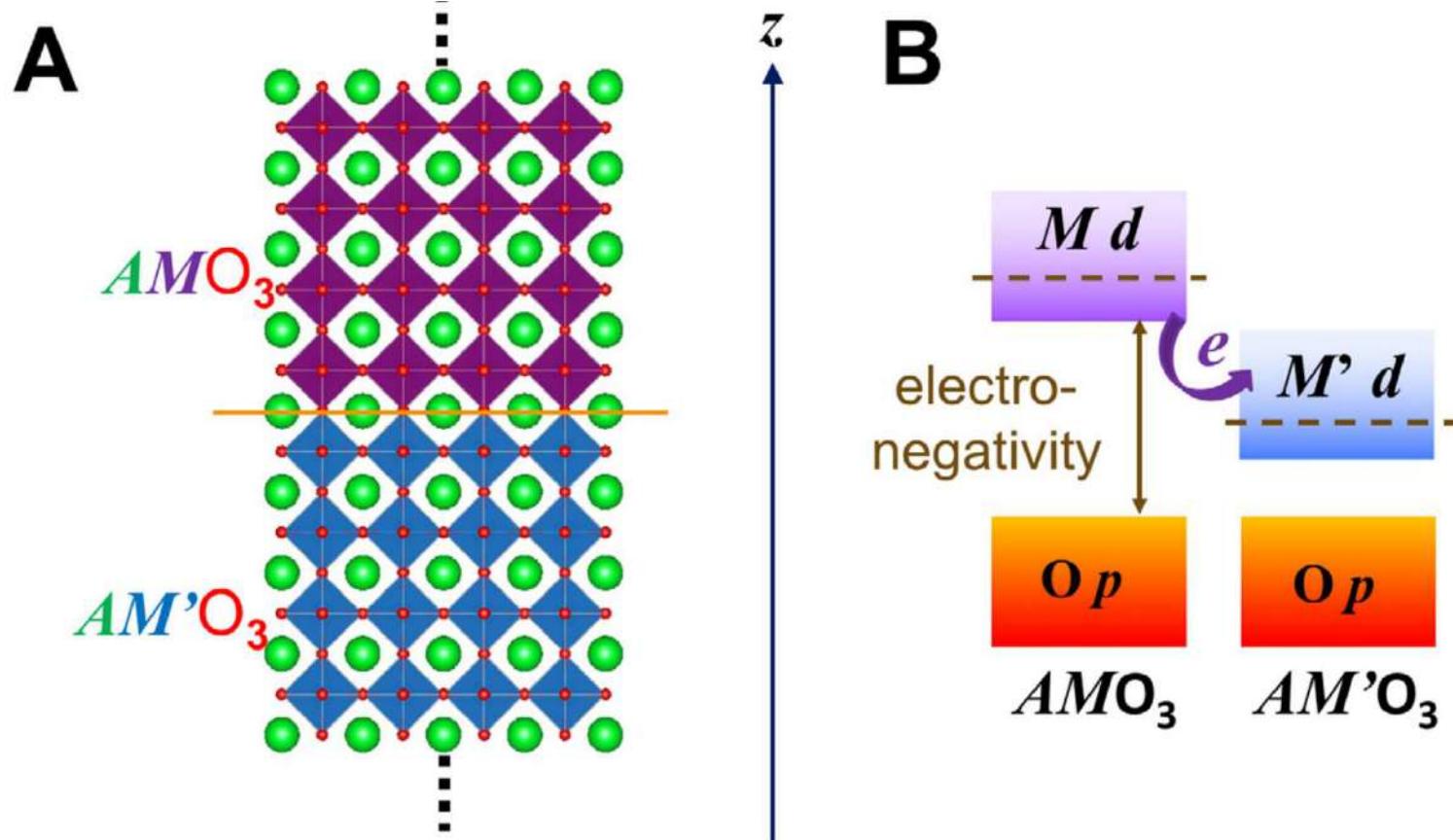
## Electronic structure of transition metal perovskites



- As the mass of transition metal elements increases, the metal d level decreases.
- For titanates, Ti-d states lie above O-p by about 3 eV.
- For nickelates and cuprates, Ni-d and Cu-d states even lie below O-p states, leading to a 'negative charge transfer' energy and strong hybridization.

Chen and Millis, JPCM, 29, 243001 (2017)

## Interfaces between oxides with partially filled d shells



- Due to continuity of oxygen lattice, oxygen p states will align
- Charge transfer will occur from material to larger electronegativity to material with smaller electronegativity

Chen and Millis, JPCM, 29, 243001 (2017)

### 1. $d_0$ -oxide-based interfaces

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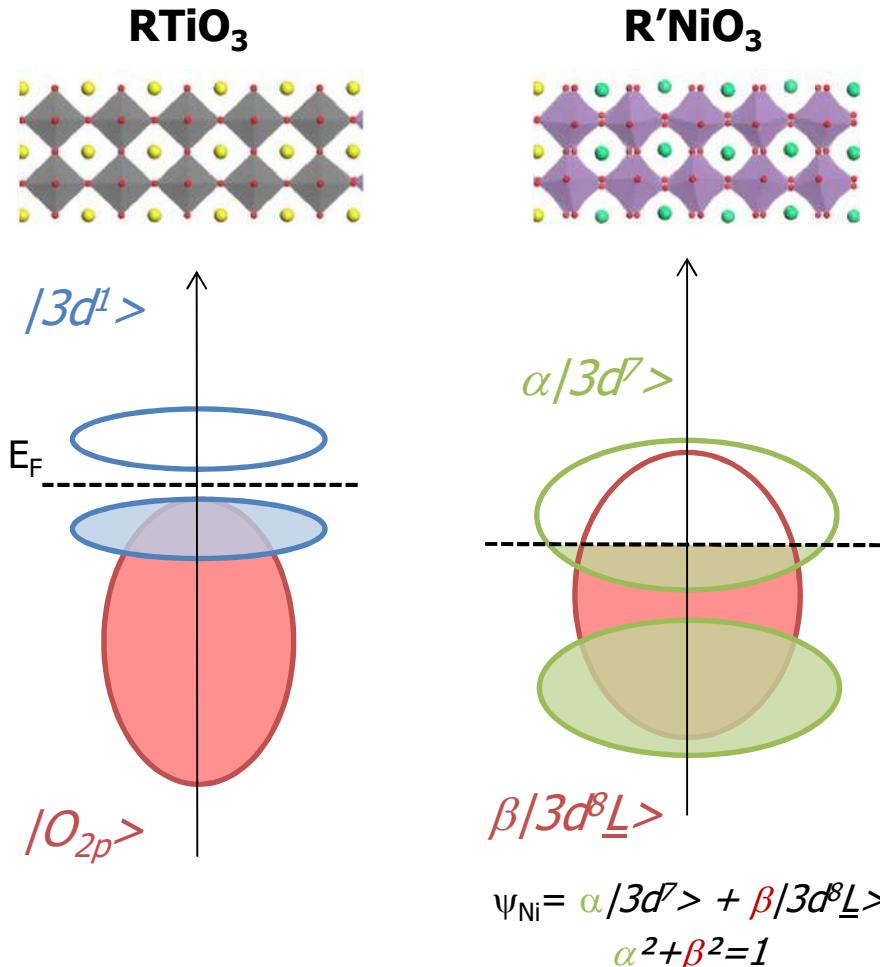
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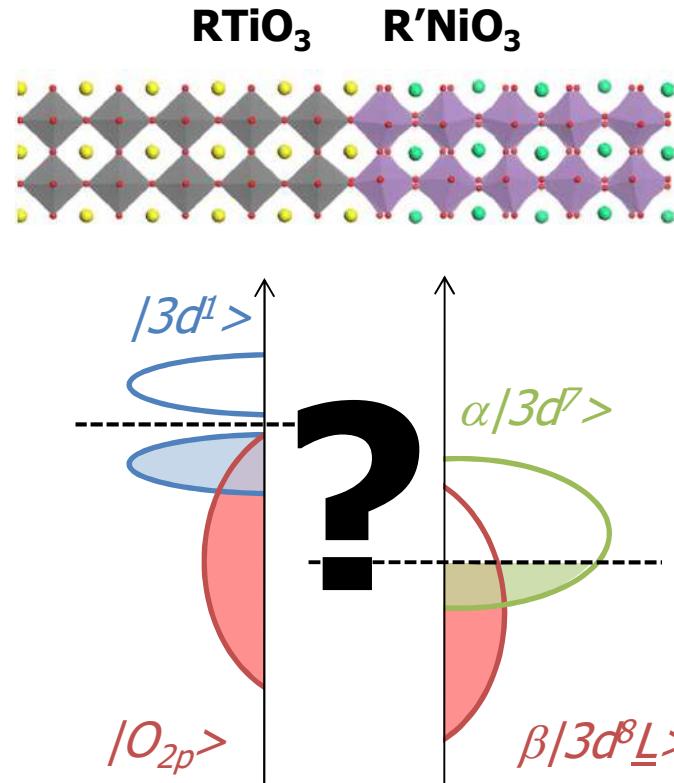
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#### 2.2 Nickelate/Titanate interfaces

# Interfaces between oxides with partially filled d shells



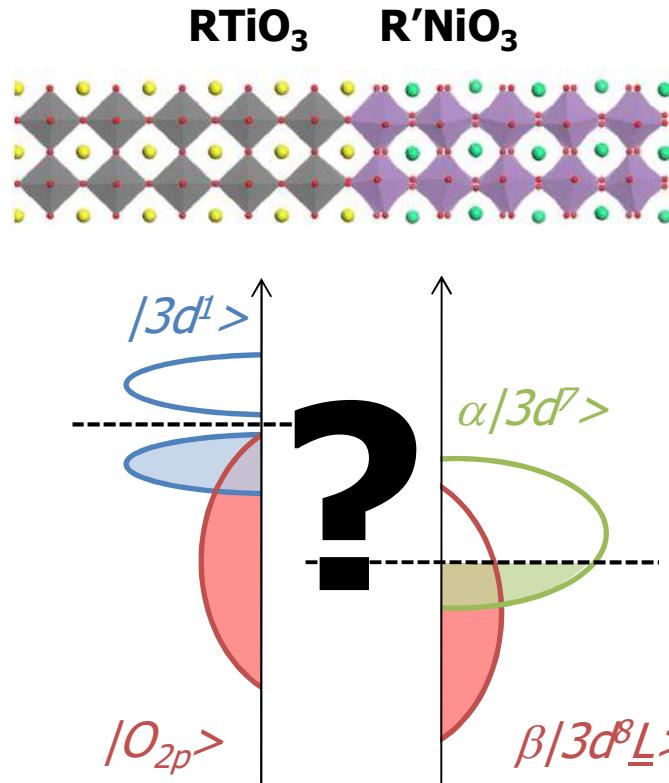
# Interfaces between oxides with partially filled d shells



**Unusual band structure** → complex electronic reconstruction at interface

**Degrees of freedom : charge, structure, orbital, spin**

# Interfaces between oxides with partially filled d shells



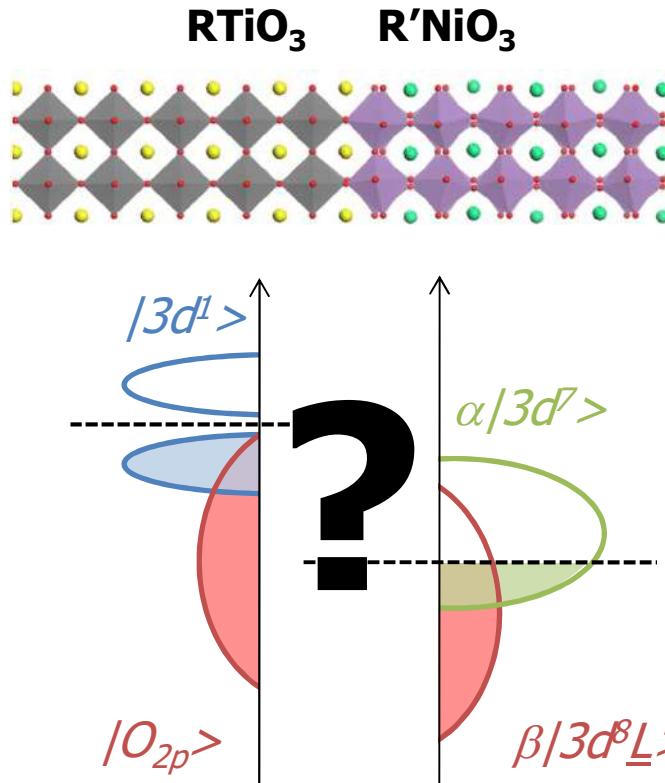
**Unusual band structure** → complex electronic reconstruction at interface

**Degrees of freedom : charge, structure, orbital, spin**

Various predicted **new properties** in related systems

- ❖ **Ferromagnetic 2-dimensional electron gases** in  $\text{LaTiO}_3/\text{SrTiO}_3$  Okamoto & Millis, Nature (2004)
- ❖ **Superconductivity** in  $\text{LaNiO}_3/\text{LaMnO}_3$  Chaloupka et al, PRL (2008)
- ❖ **New spin/orbital states** in  $\text{GdTiO}_3/\text{SrTiO}_3$  Li et al, ArXiv (2013) ; in  $\text{LaTiO}_3/\text{LaNiO}_3$  Millis et al, PRL (2013)
- ❖ **New ferroelectrics** in  $\text{LaGaO}_3/\text{YGaO}_3$  Rondinelli & Fennie, Adv. Mater. (2012)

# Interfaces between oxides with partially filled d shells



**Unusual band structure** → complex electronic reconstruction at interface

**Degrees of freedom : charge, structure, orbital, spin**

- ❖ Here systematic study of interface properties between  $\text{GdTiO}_3$  and  $\text{RNiO}_3$
- ❖ Change rare-earth (La, Nd, Sm) : **tune bandwidth, bond angles, p-d energy**, etc

# Interfaces between titanates and nickelates

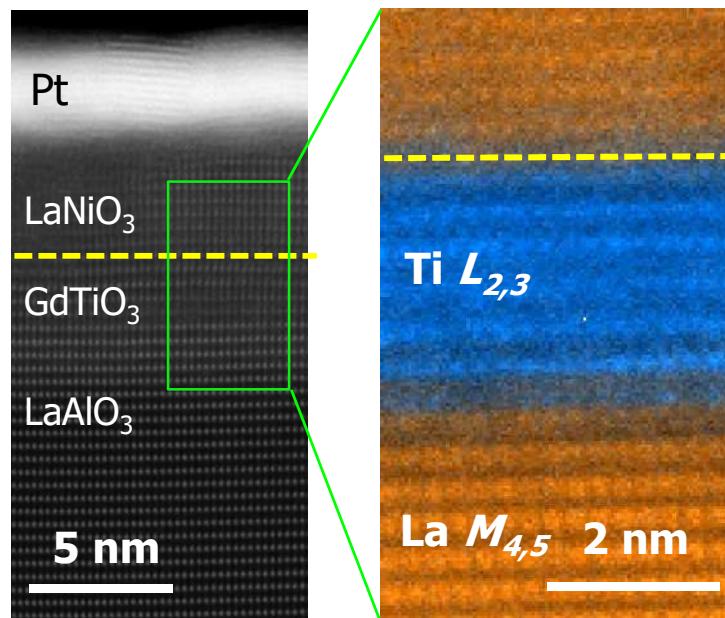
## Electronic reconstruction at $RNiO_3/GdTiO_3$ interfaces



- ➊ Epitaxial growth of bilayer system

# Interfaces between titanates and nickelates

## Electronic reconstruction at $R\text{NiO}_3/\text{GdTiO}_3$ interfaces

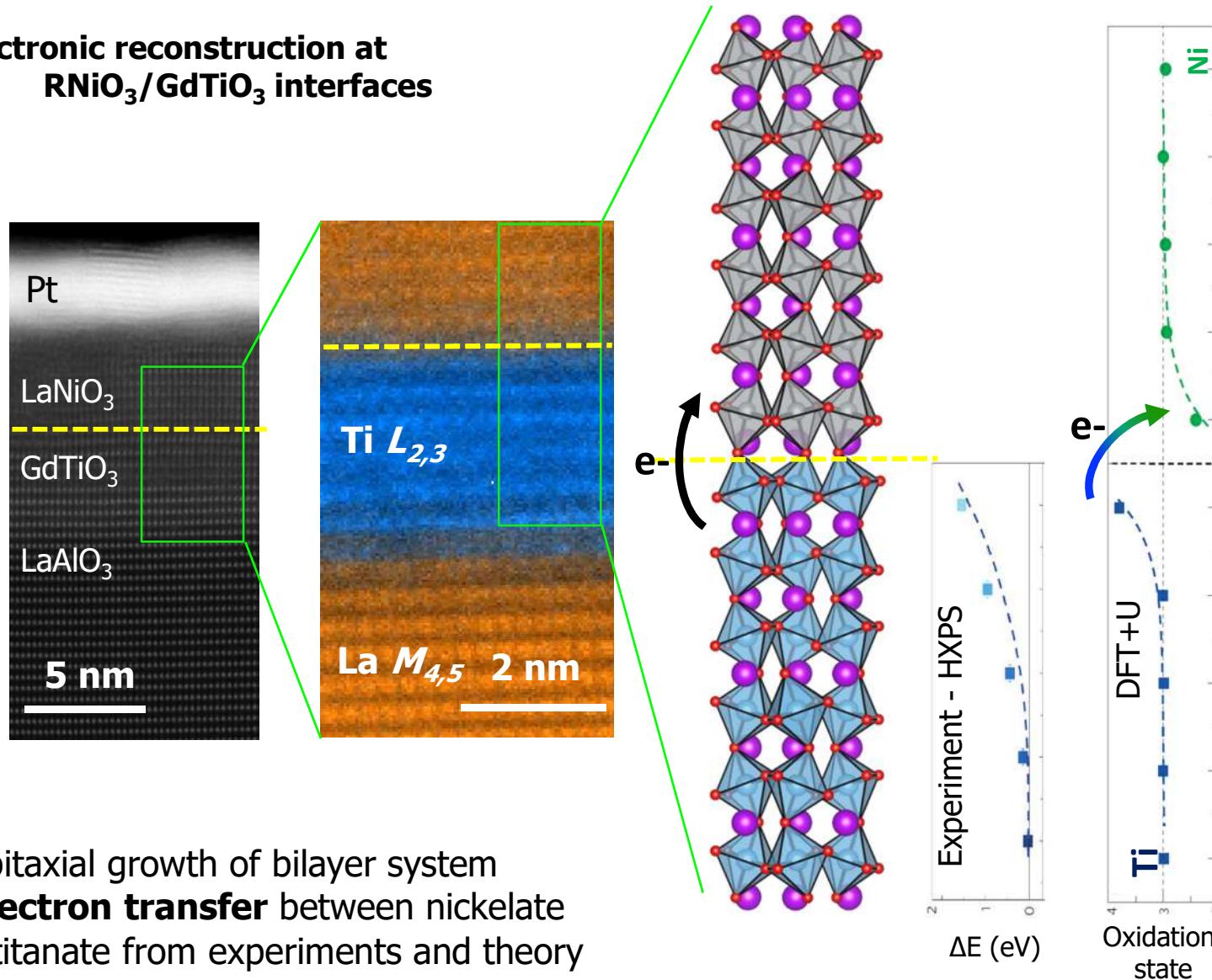


- Epitaxial growth of bilayer system

Grisolia, MB et al, Nature Phys. 12? 484 (2016)

# Interfaces between titanates and nickelates

## Electronic reconstruction at R<sub>x</sub>NiO<sub>3</sub>/GdTiO<sub>3</sub> interfaces

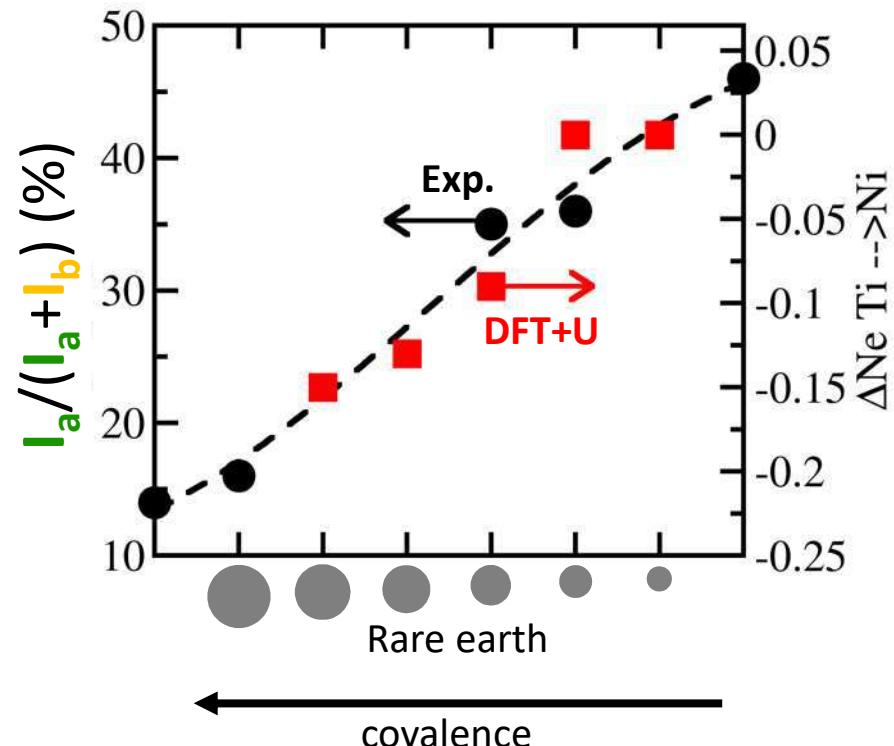
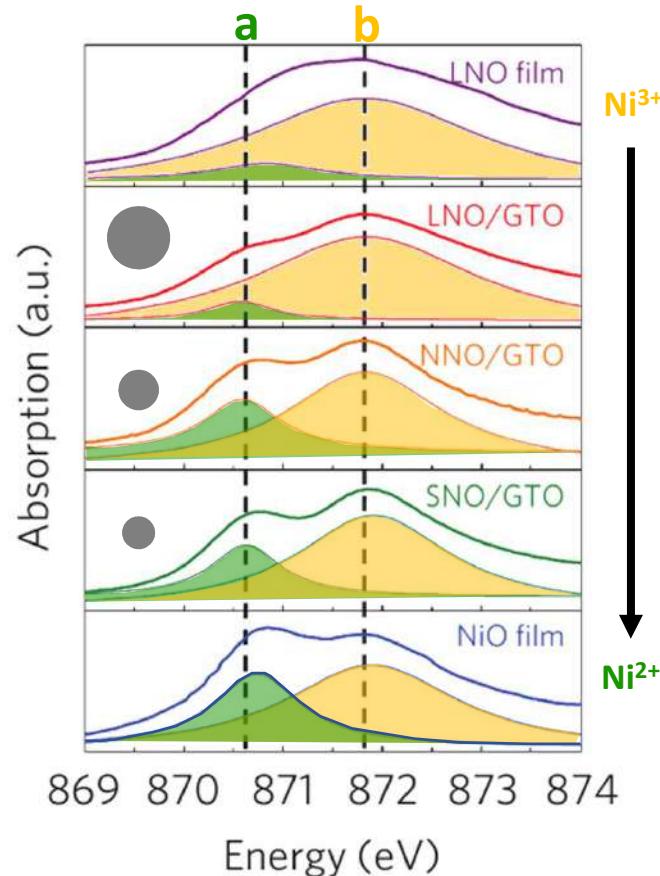


Grisolia, MB et al, Nature Phys. 12? 484 (2016)

# Interfaces between titanates and nickelates

## Electronic reconstruction at $\text{RNiO}_3/\text{GdTiO}_3$ interfaces : role of covalence

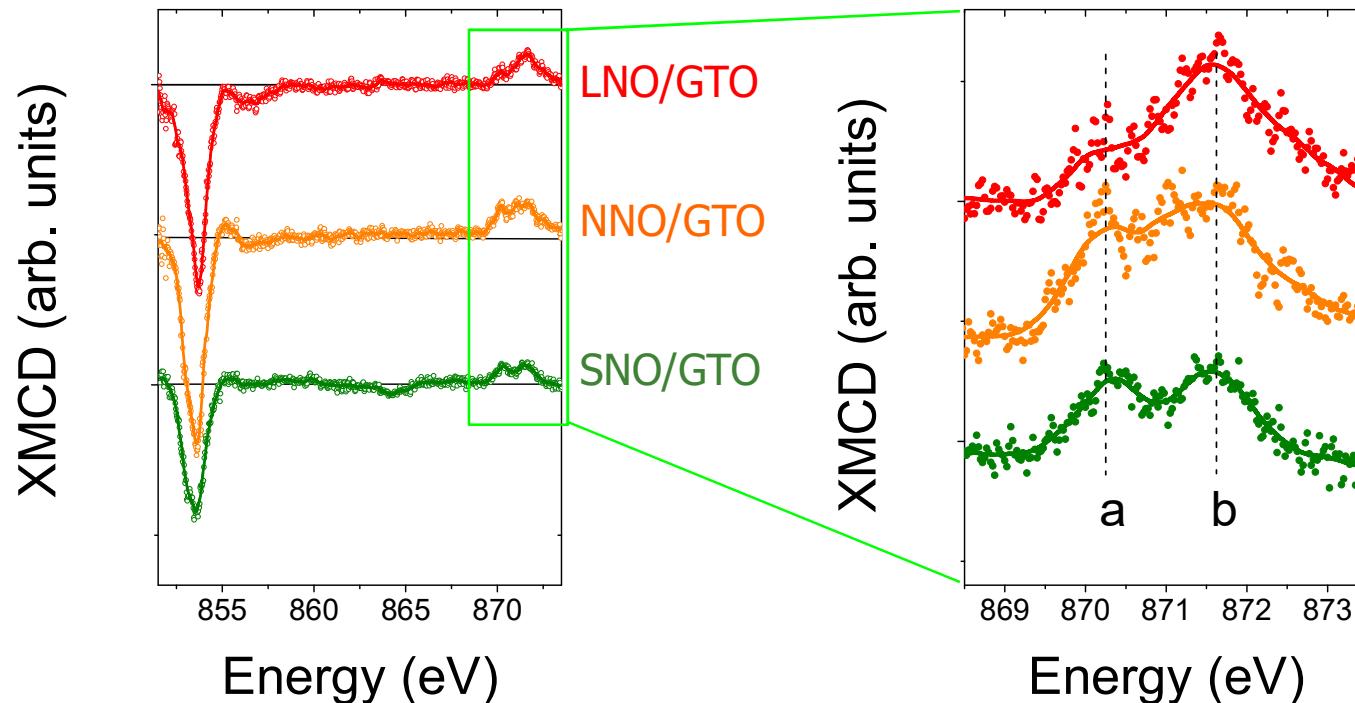
Exp. : XAS - Ni  $L_3$  edge



- Amount of charge transferred from titanate to nickelate **modulated by covalence** in nickelate material

Grisolia, MB et al, Nature Phys. 12? 484 (2016)

## Magnetic properties of $\text{RNiO}_3/\text{GdTiO}_3$ interfaces



- XMCD is observed at Ni  $L_{2,3}$  edge for all bilayers while individual nickelates are AF
- Strong signature of **Ni<sup>2+</sup> magnetism**
- « Covalent exchange » can induce ferromagnetic coupling between Ni<sup>2+</sup> and Ni<sup>3+</sup>
- **Novel magnetic state in nickelates**

Goodenough, PR 100, 554 (1955)

Goodenough, J. Solid State Chem. 127, 126 (1996)

Grisolia, MB et al, Nature Phys. 12? 484 (2016)

## Conclusions and perspectives

- Oxide interfaces have **unexpected electronic and magnetic properties**
- Some properties derive from the bulk of the compounds involved, **some are readily new**
- **Inversion symmetry breaking** is key to most new properties
- Most literature focuses on  $\text{SrTiO}_3$  and related  $d^0$  compounds like  $\text{KTaO}_3$
- Much work remains to be done for **interfaces between non  $d^0$  perovskites**
- Harder to grow as interesting couples often need to be grown in different conditions
- Parameter space is huge and more **exotic phenomena** should arise from orbital and spin reconstruction (**topological effects**)

MB et al, Adv. Phys. 60, 5 (2011)

J. Varignon, MB et al., Nature Phys. 14, 322 (2018)